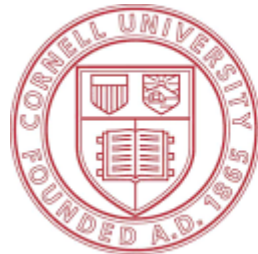


# Cathode Characterization and Fabrication

Applying the tools of modern material science to cathode design

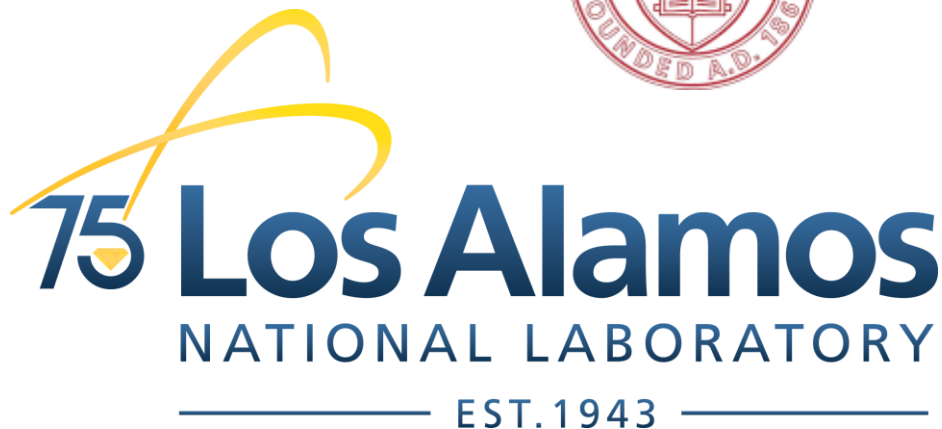


Cornell University

John Smedley

Snowmass2021 Electron Source Workshop

February 16, 2022



**BROOKHAVEN**  
NATIONAL LABORATORY

Thanks to Mengjia Gaowei and Alice Galdi (and many others)



Operated by Los Alamos National Security, LLC for the U.S. Department of Energy's NNSA

# Talk Outline

- **Introduction – what makes a good photocathode?**
  - Focus on Alkali Antimonides and Tellurides
- **Deposition methods**
- **Characterization**
- **Path forward**

# Photocathode needs in accelerator applications



**Electron beam required  
for e-cooling**

## **FEL sources**

Moderate currents  
Emittance improvement  
(ideally  $0.1 \mu\text{m}/\text{mm}$ )



High average current ( $> 100 \text{ mA}$ )

High bunch charge ( $1 \text{ nC}$ )

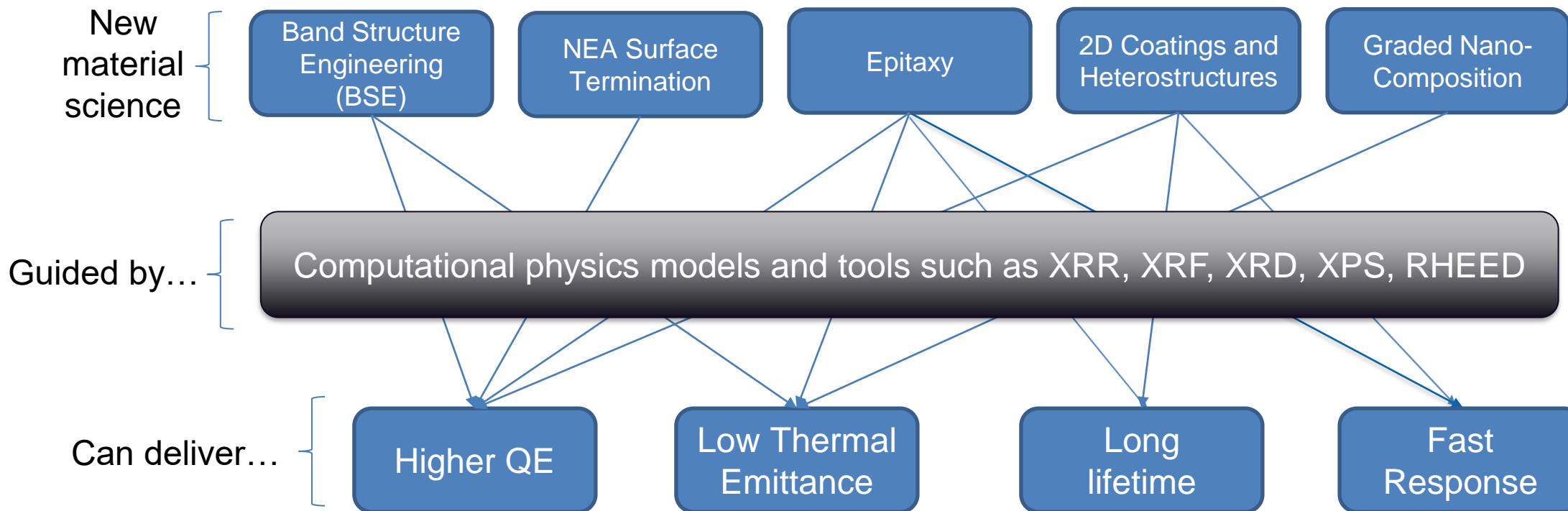
Long lifetime ( $> 1 \text{ week}$ )

Reproducible

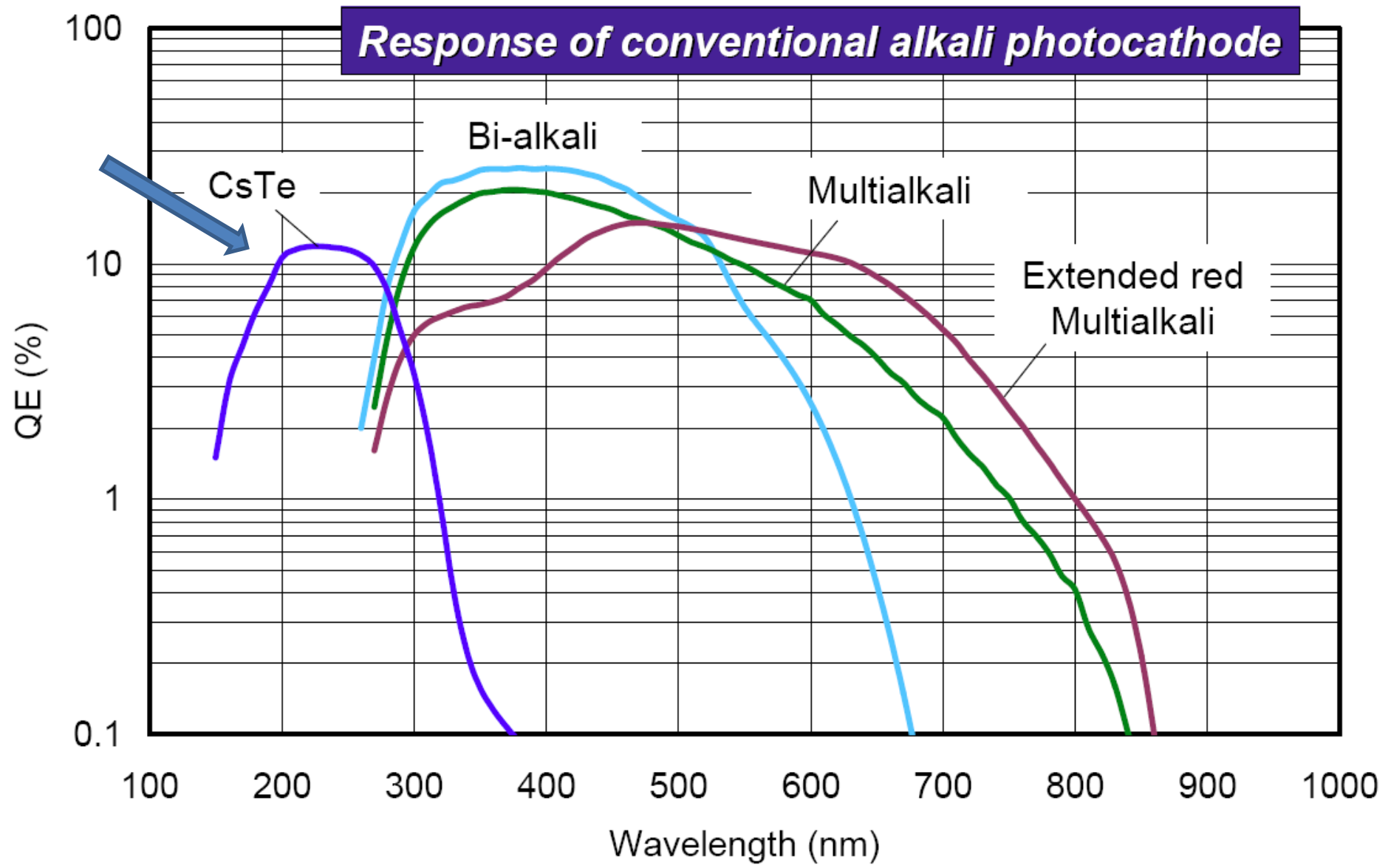
## **Ultrafast Electron Diffraction/Microscopy**

High brightness  
Very low current  
Short pulse duration

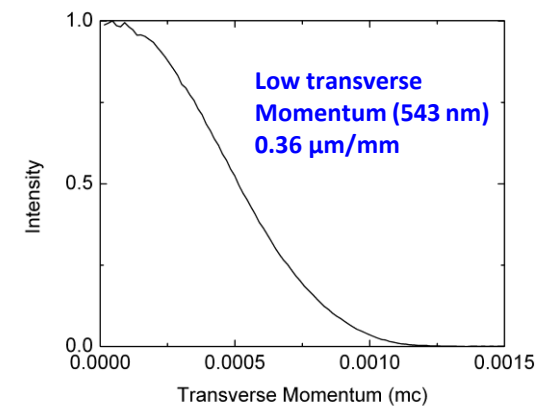
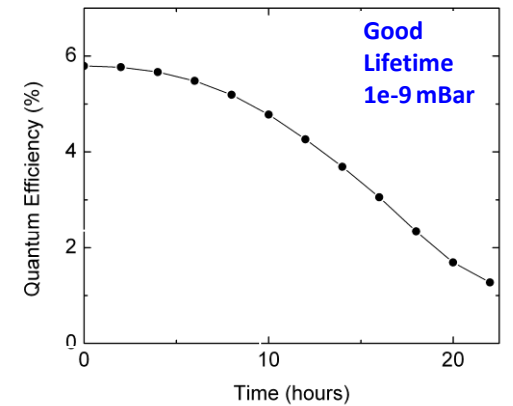
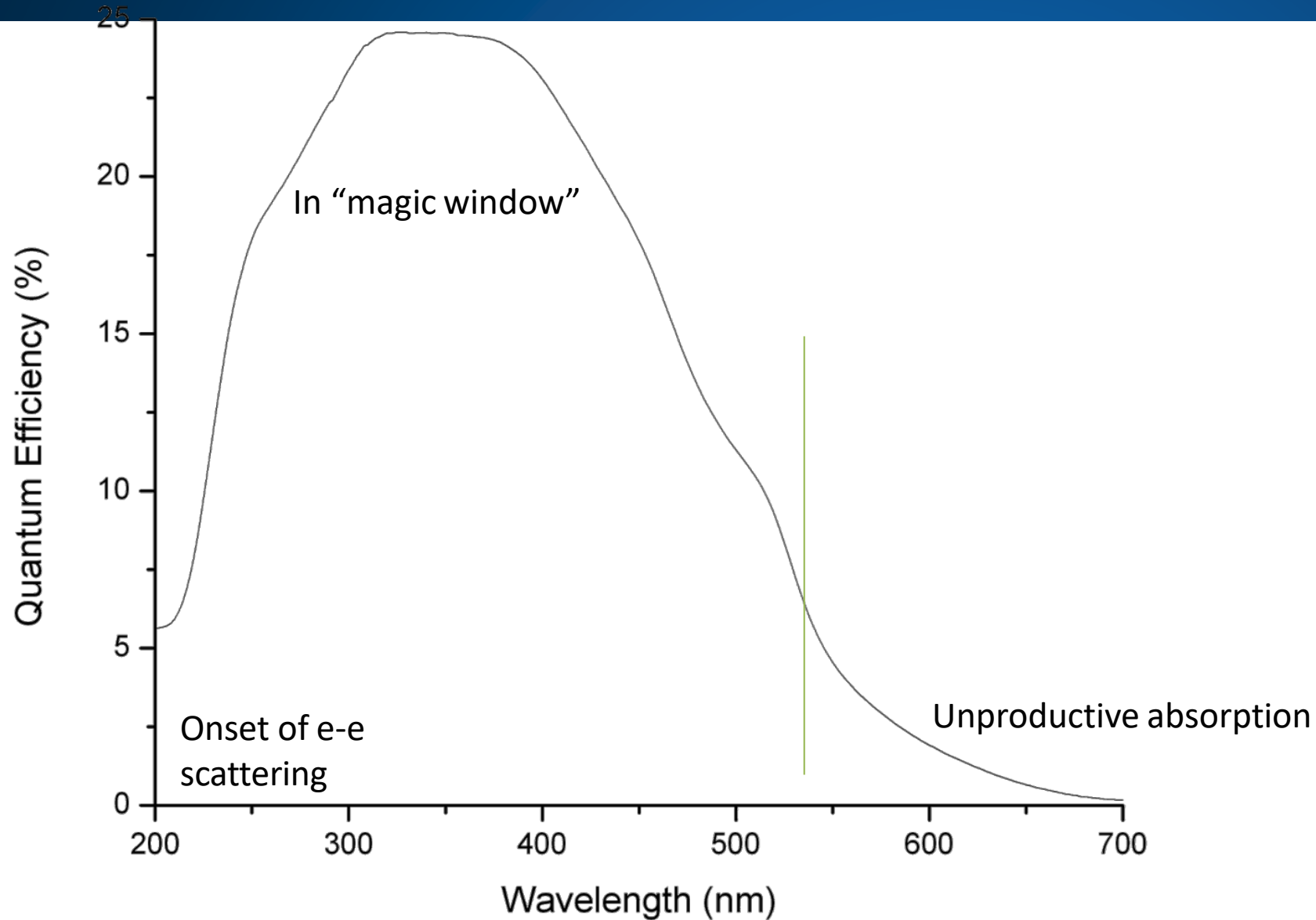
# New fabrication techniques are developed and applied to photocathodes motivated by theory and guided by real-time *in situ* x-ray analysis



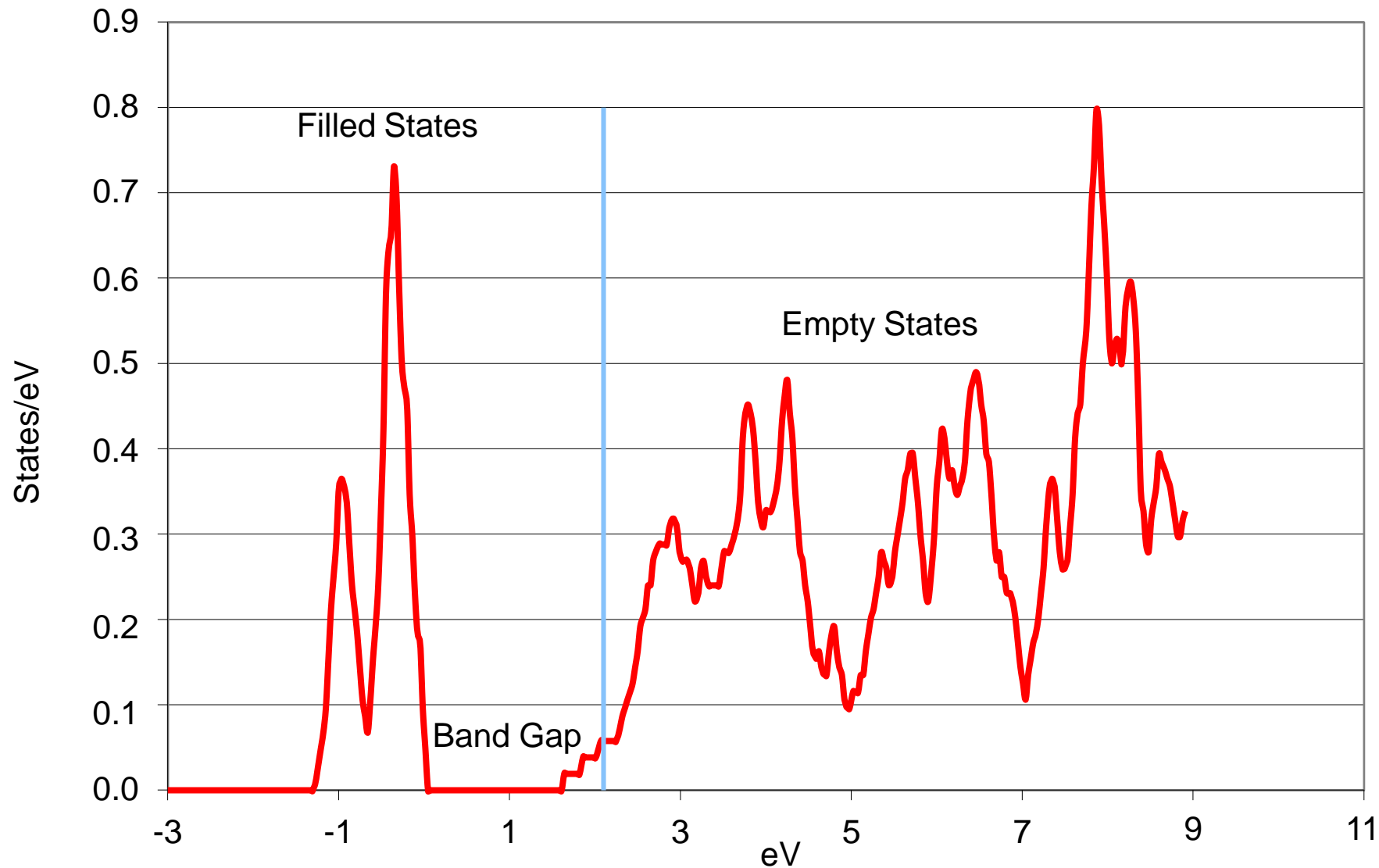
- *In Situ* or rapid feedback is required to optimize for material properties other than QE
- Modeling is needed to guide growth and understand properties



# K<sub>2</sub>CsSb: A Good Candidate



# $K_2CsSb$ Density of States



Advantages of the Antimonides

- Narrow Valence Width
- High Optical Density
- Fast (for semiconductors)

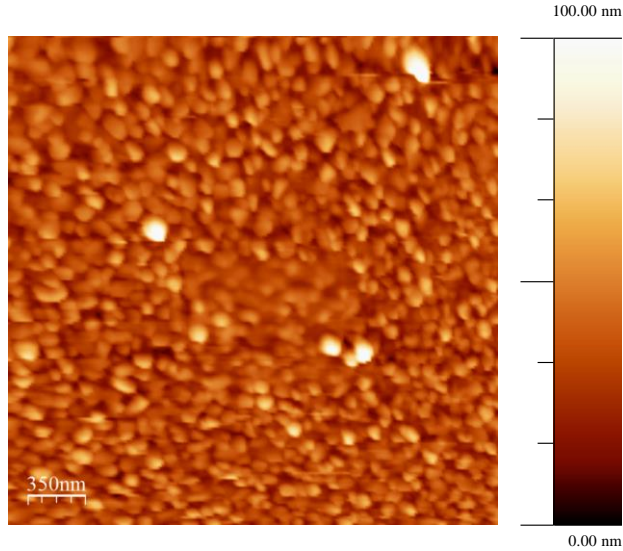
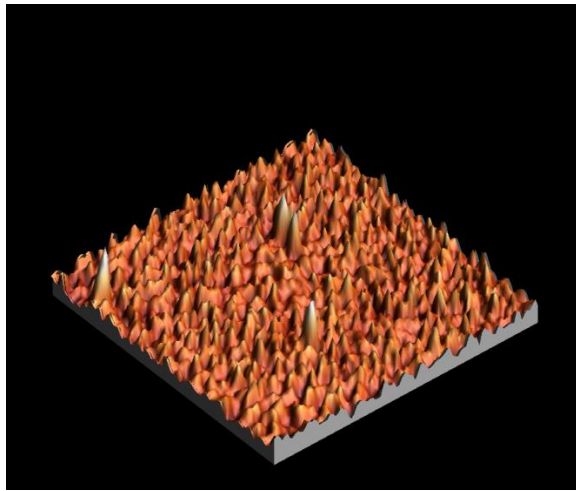
# Deposition Methods

- **Sequential deposition via thermal evaporation**
  - Easy to control (one element at a time)
  - Recipe to maximize QE
- **Co-evaporation via thermal evaporation or effusion cells**
  - Much harder to control stoichiometry
  - Potentially much better films
- **ALD – difficult as no alkali precursors**
- **Sputtering and Pulsed Laser Deposition**
  - Difficult to control stoichiometry, does not sputter stoichiometricly
  - Bulk targets are a pain (require custom fabrication and vacuum load lock)
  - Can fix stoichiometry with Cs evaporation: M. Gaowei, et al., Synthesis and x-ray characterization of sputtered bi-alkali antimonide photocathodes. *APL Materials*, 2017. 5(11): p. 116104
  - PLD of Sb can lead to precision control of deposition – Digital Growth!



# Traditional Sequential Deposition of K<sub>2</sub>CsSb

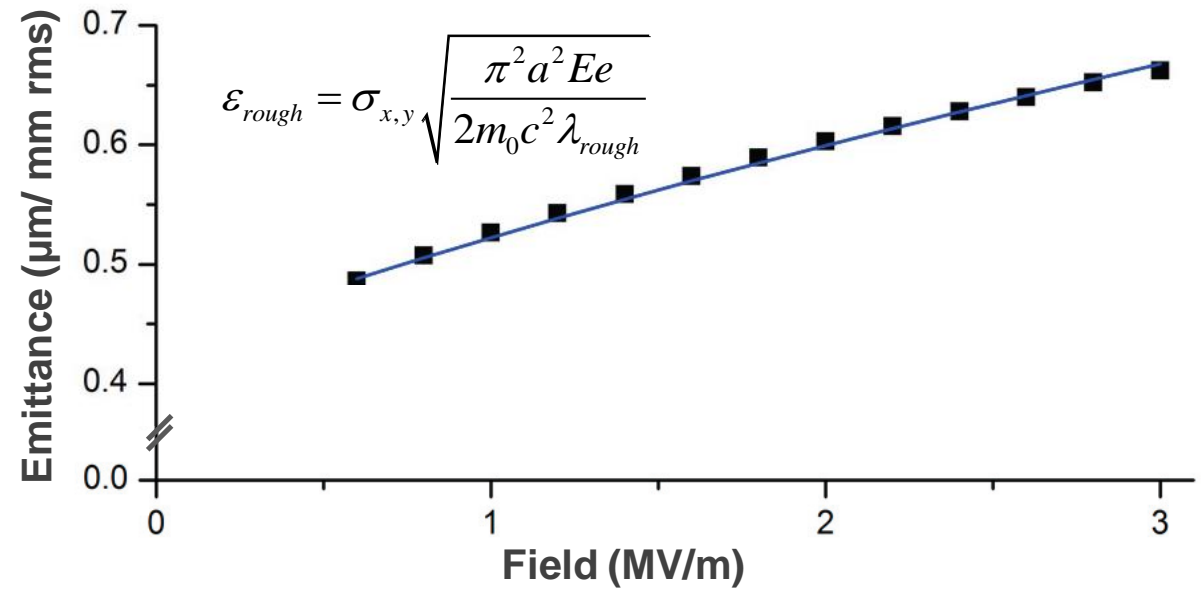
## High QE and Rough Surface



S. Schubert et al., APL Materials 1, 032119 (2013)

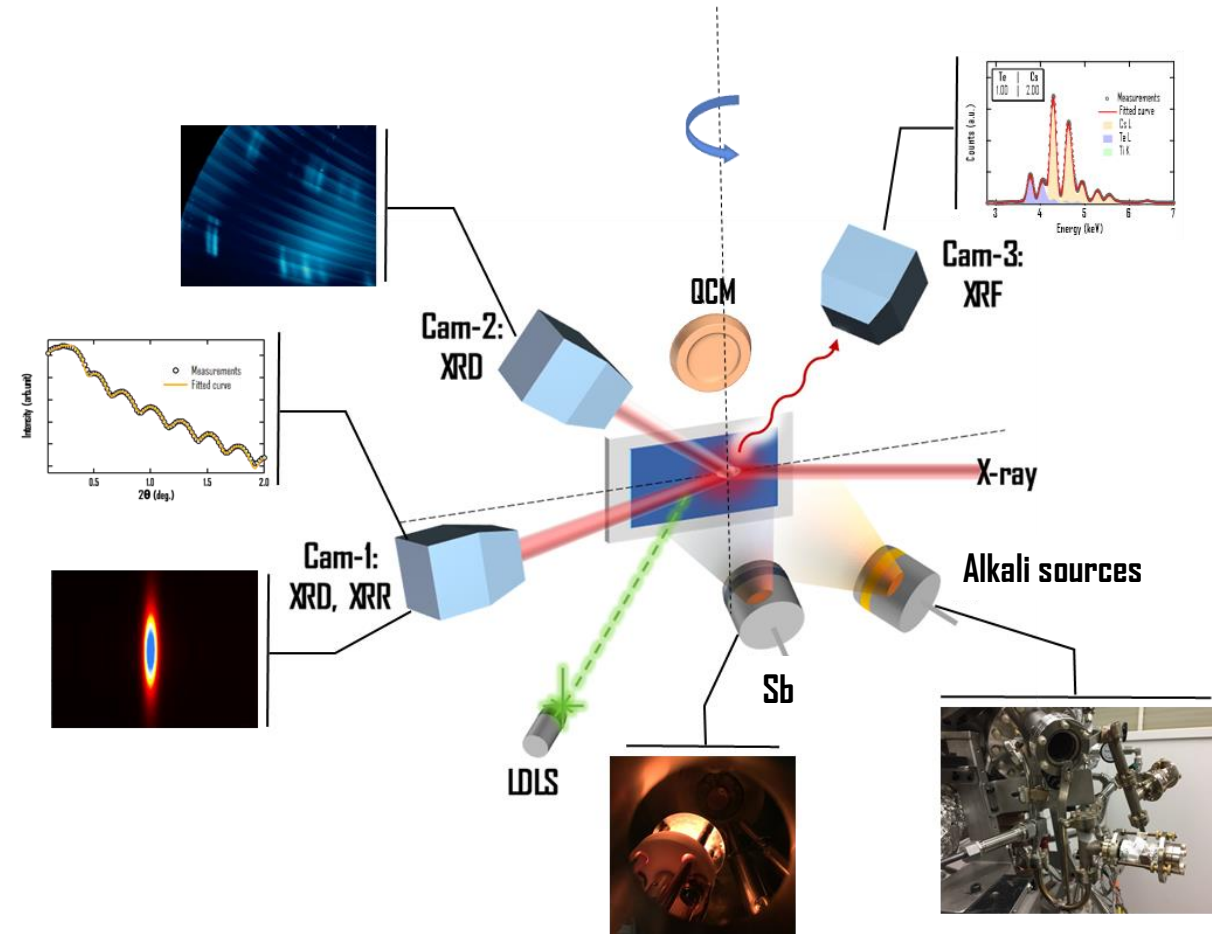
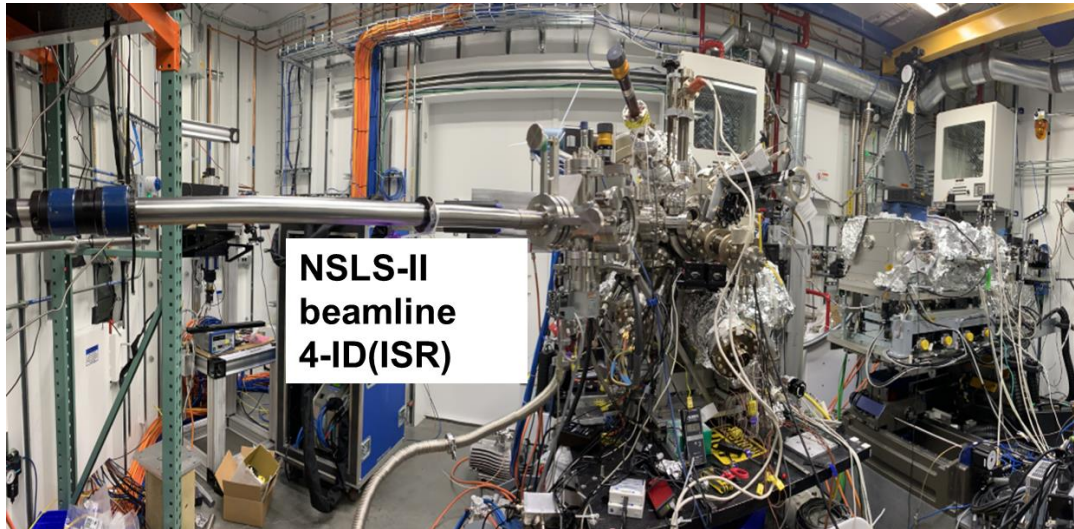
**Emittance vs field  
measured with  
Momentatron, 532 nm light**

**25 nm roughness,  
100 nm spatial period**



T. Vecchione, et al, Proc. of IPAC12, 655 (2012)

# Cathode Material development @BNL : In situ and real time x-ray characterization



## Growth controls:

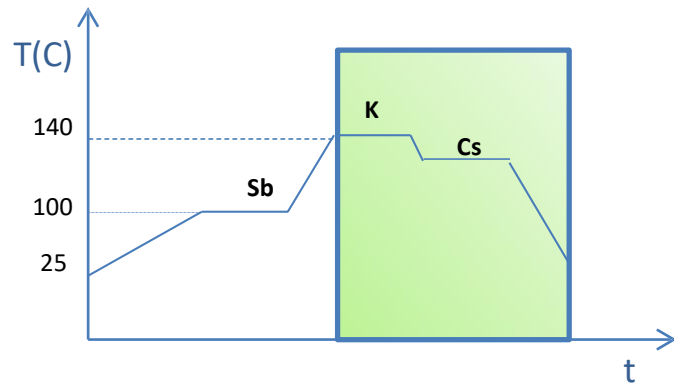
- $T_{\text{sub}}$
- Flux rate

## Characterization:

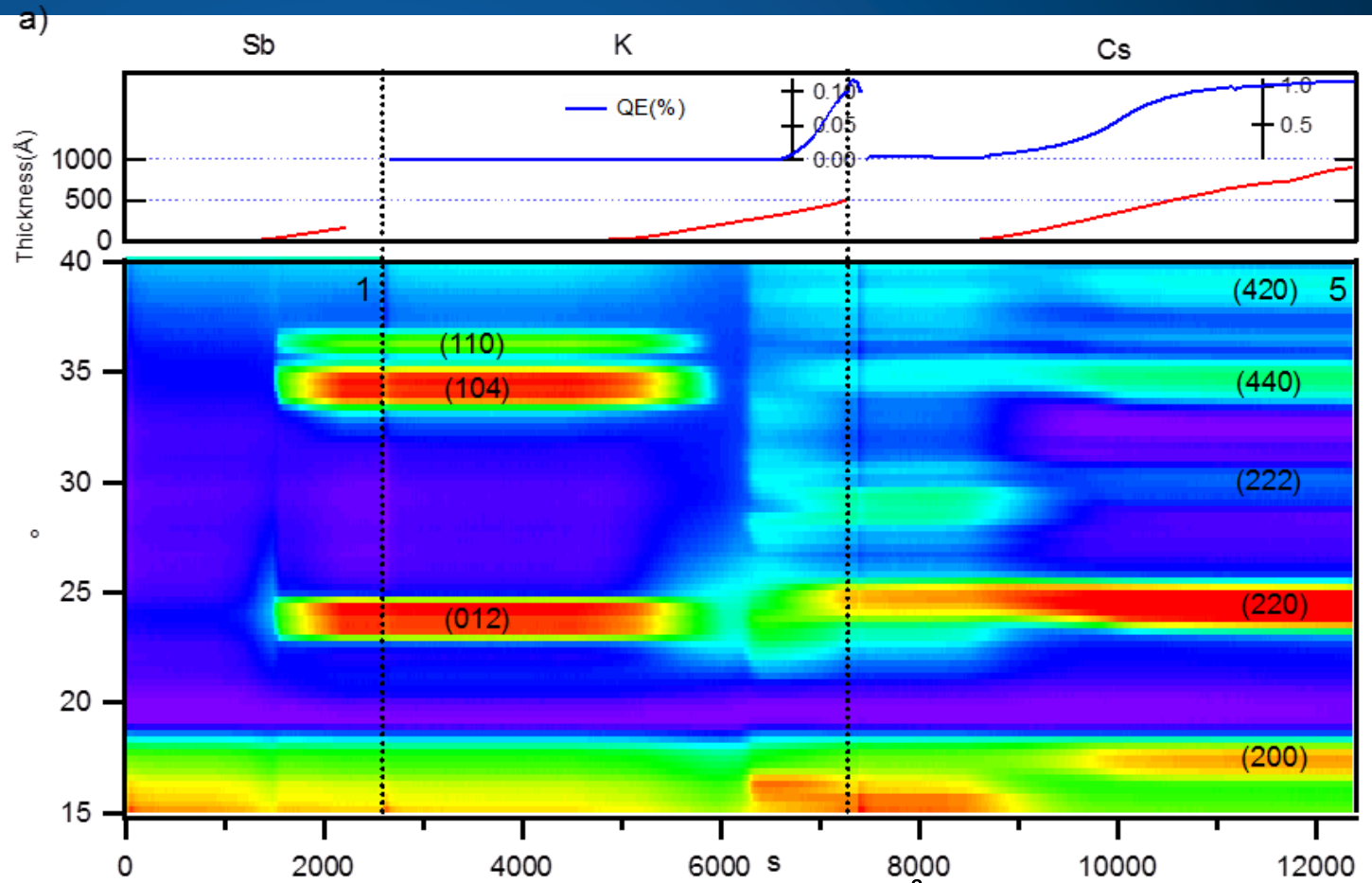
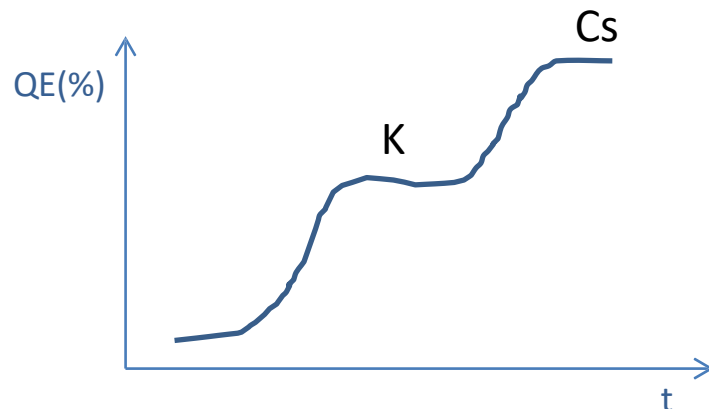
- QCM
- XRD
- XRR
- XRF
- QE

# Sequential Evaporation Reaction Dynamics

Recipe:



QE during growth (532 nm laser)

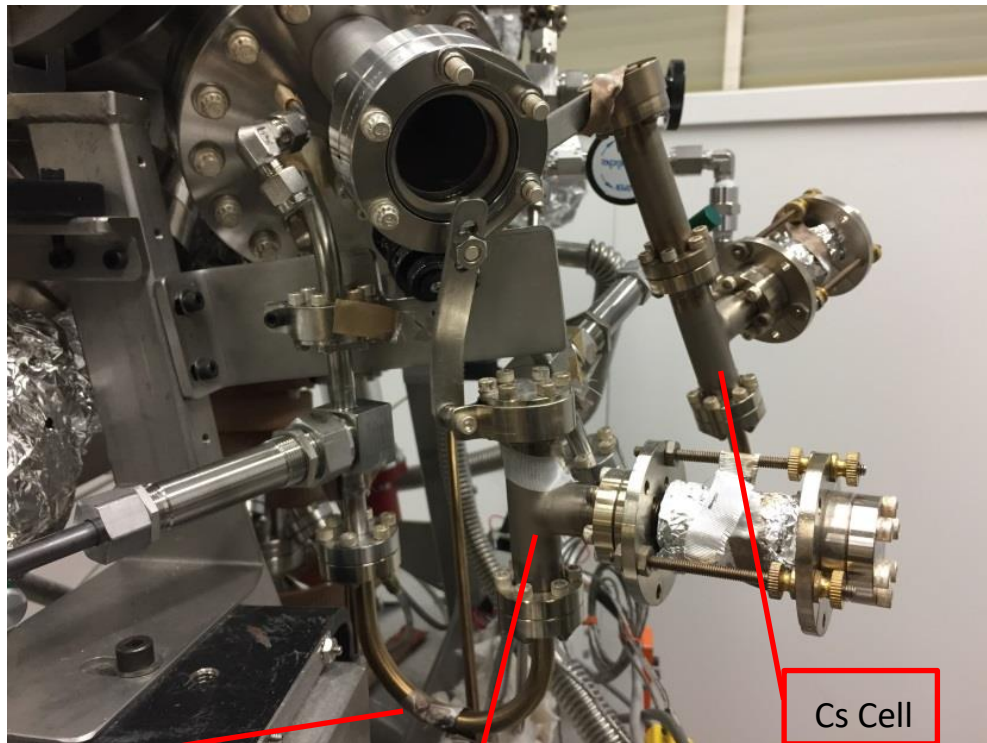


Antimony evaporated on Si, 0.2 Å/s; crystallize at 4nm  
 K deposition dissolves Sb layer - This is where roughening occurs!  
 QE increase corresponds with  $K_xSb$  crystallization  
 Cs increases lattice constant and reduces defects



# Ternary Co-evaporation

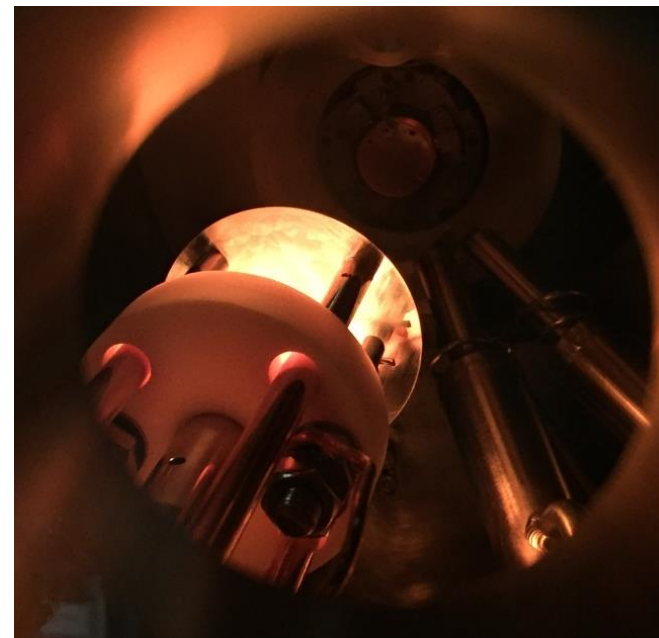
Simultaneously evaporate from Sb evaporator and K,Cs effusion cells



J tube

K capsule

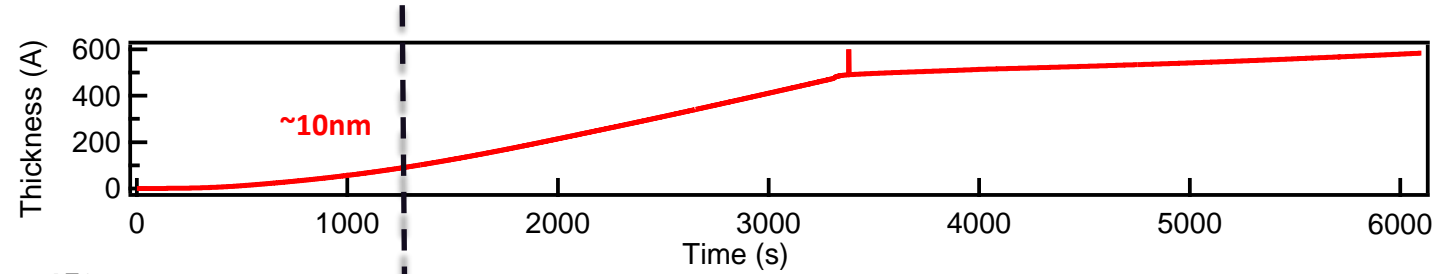
Cs Cell



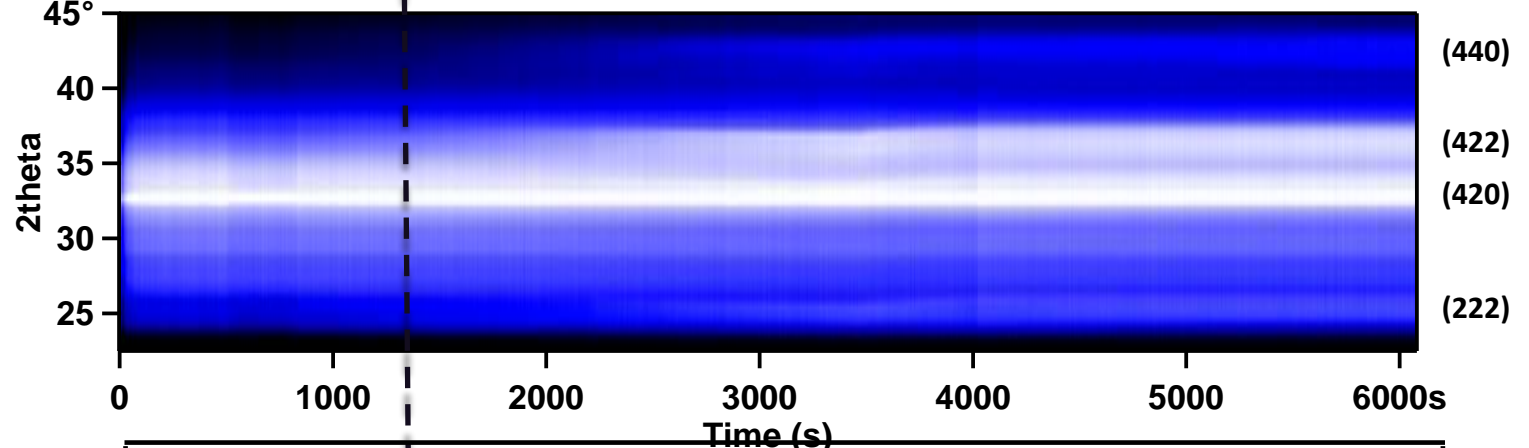
Growth rate are controlled by J tube temperature, valve and shutter

Stoichiometry controlled by real time XRF

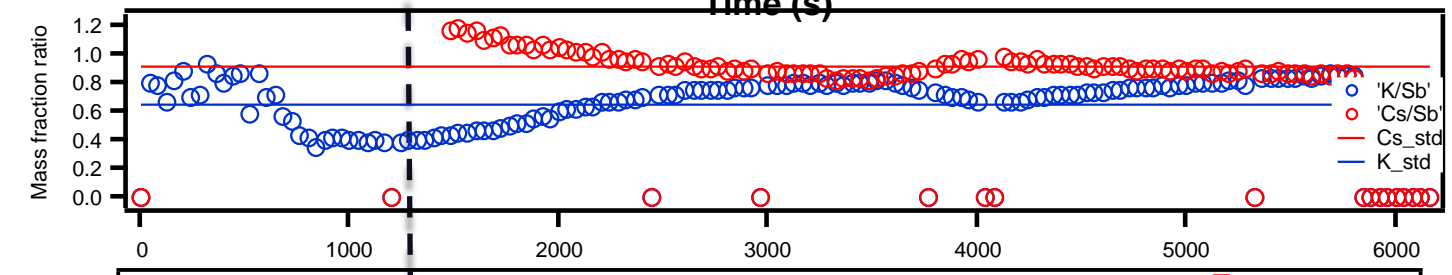
Thickness



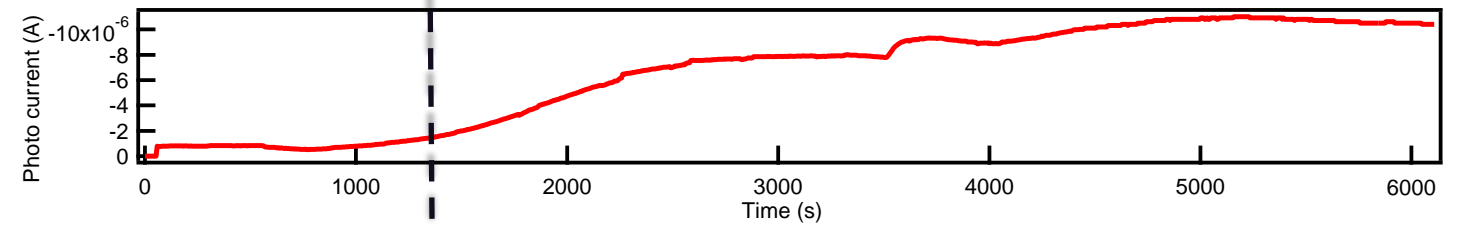
Real time XRD



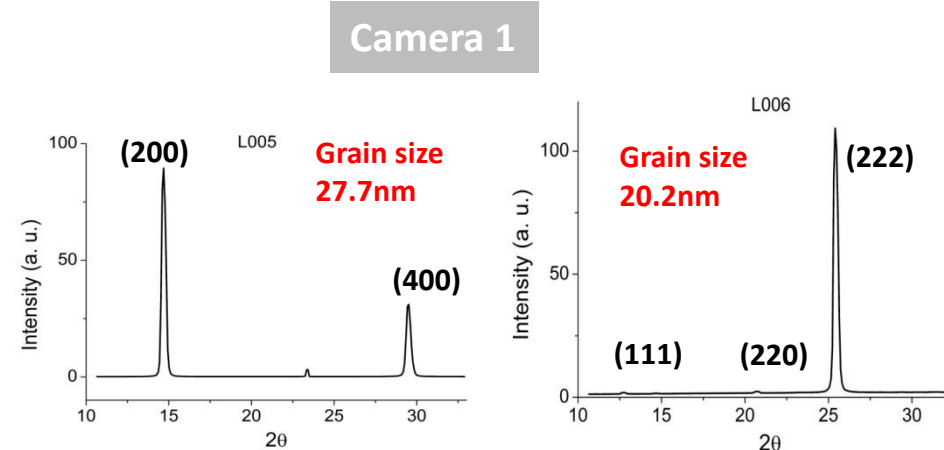
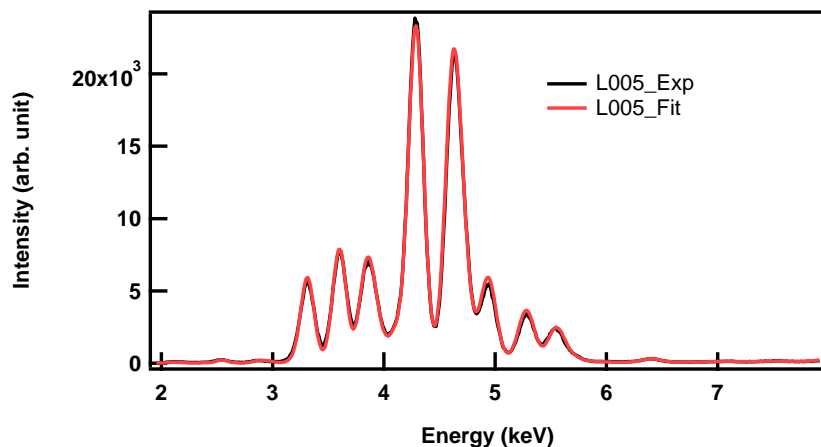
Real time Fluorescence



QE

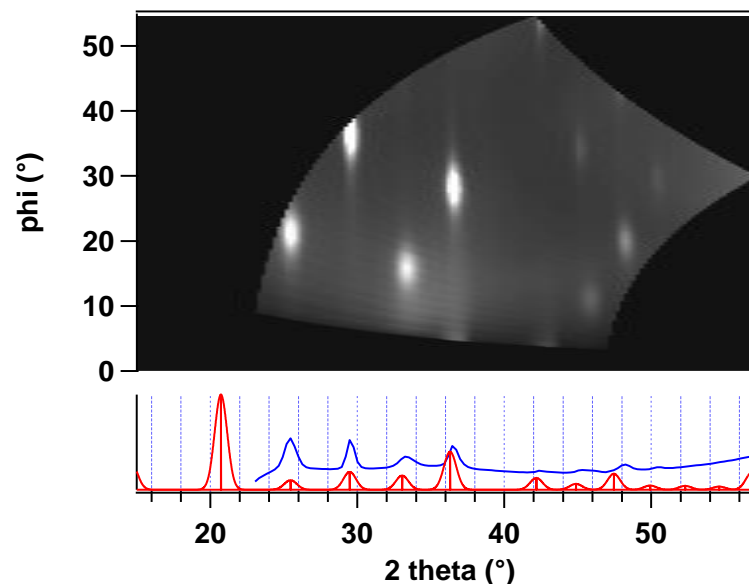


# Stoichiometry & Structural Analysis



	K	Sb	Cs
L004 Si	2.50	1.00	1.16
L005 Si	2.37	1.00	0.91
L006 Si	2.21	1.00	0.95
L011 Si	2.07	1.00	0.94
L012 MgO	1.98	1.00	0.88

**Good K/Cs/Sb ratio!**

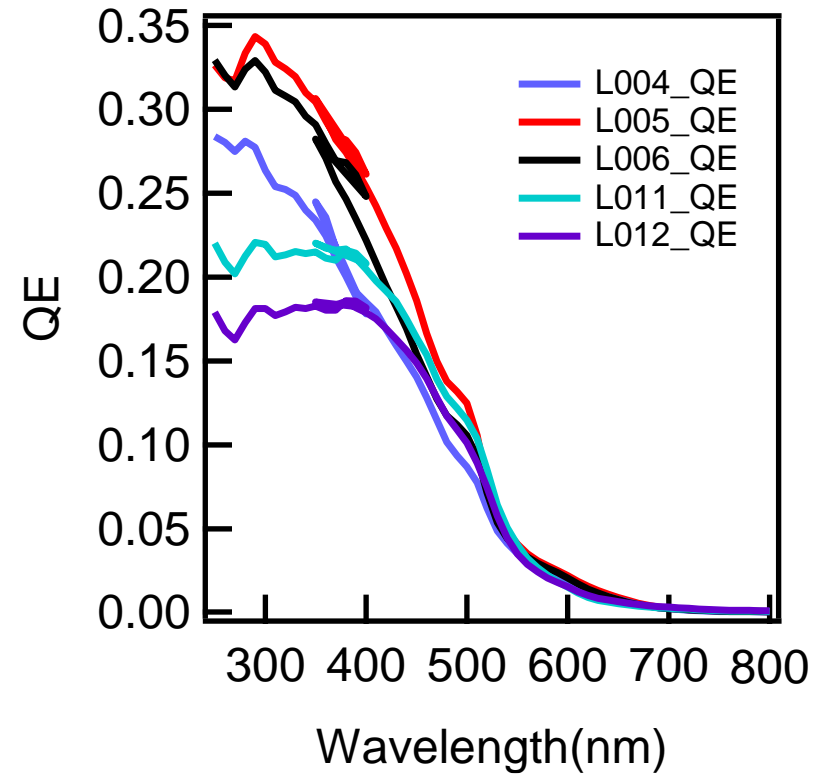
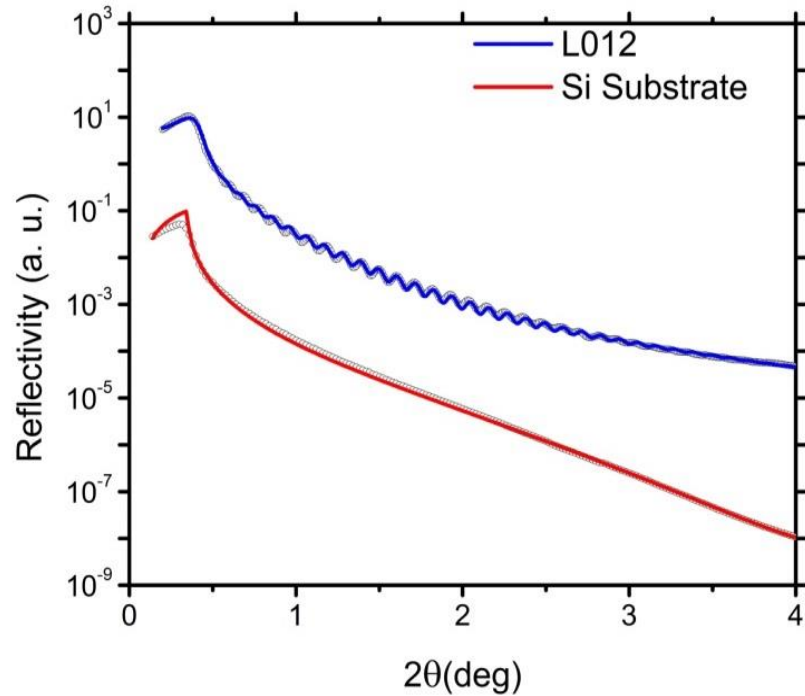


**Highly textured  $K_2CsSb$  phase!**

**mm size lateral grains**

**Works for the entire Alkali antimonide family – we've created highly textured films with a wide range of stoichiometries**

# Surface Roughness & QE



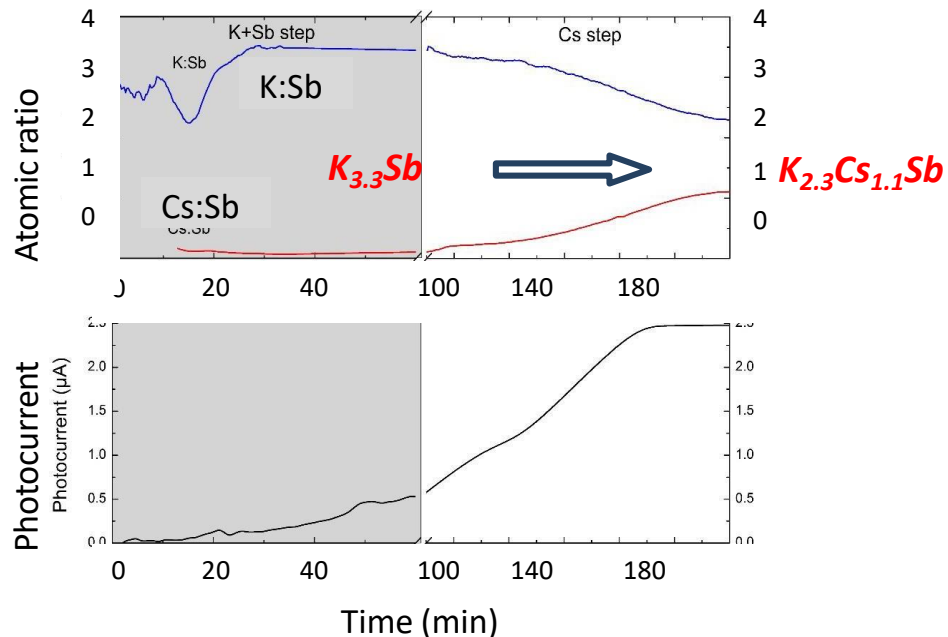
	QE @532nm(%)	Roughness(A)	Thickness (A)	Grain size (A)
L004 Si	4.9	3.5	234	155
L005 Si	5.8	11.5	815.3	277
L006 Si	5.4	13.8	757.5	202

Simultaneous evaporation of all constituents results in no crystal phase transformation  
 Smooth, reproducible and **ultra-high QE**. Highly Crystalline!

# 2-step Co-evaporation

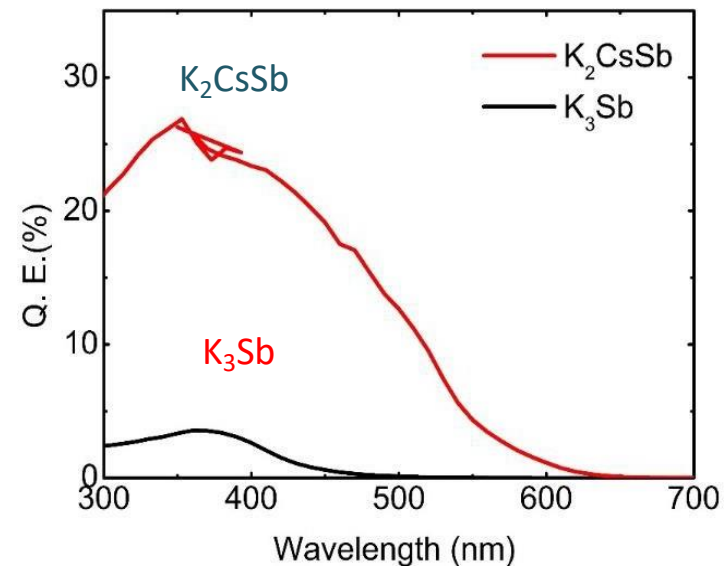
Two-step recipe:

1. K+Sb co-deposition, maximize QE at 380 nm, @ ~120°C
2. Cs deposition on top @ ~100°C until QE (530 nm) maximizes



K:Sb becomes close to 3:1 shortly after K-Sb co-deposition starts

Spectral response:



- $\text{K}_3\text{Sb}$  layer: peak 3.5% at 360 nm; 0.047% at 530 nm
- After Cs: peak 26% at 360 nm; 7% at 530 nm

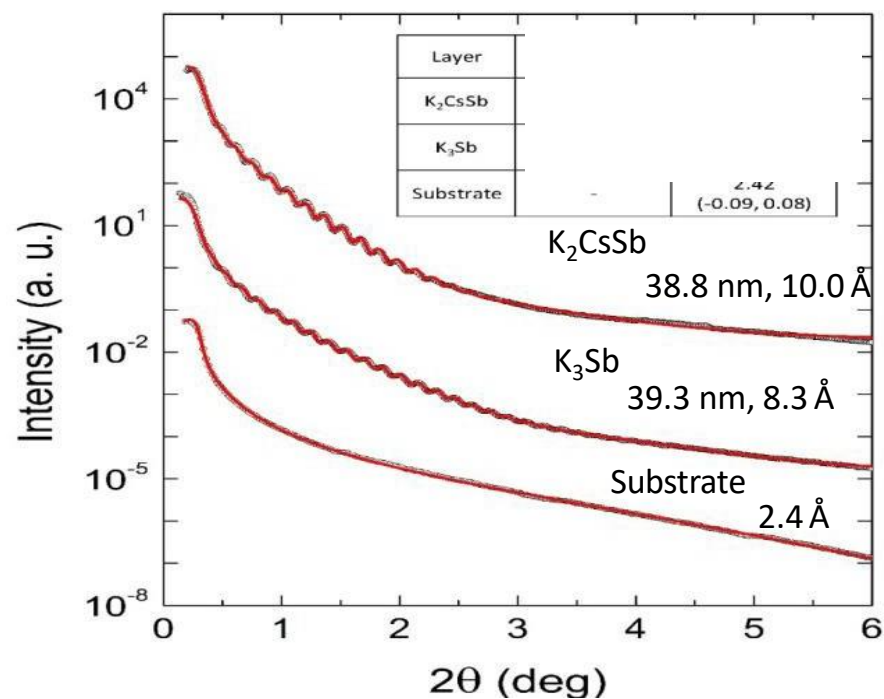
**Most of the performance advantages of Ternary co-evaporation, but MUCH easier**



# 2-step Co-evaporation

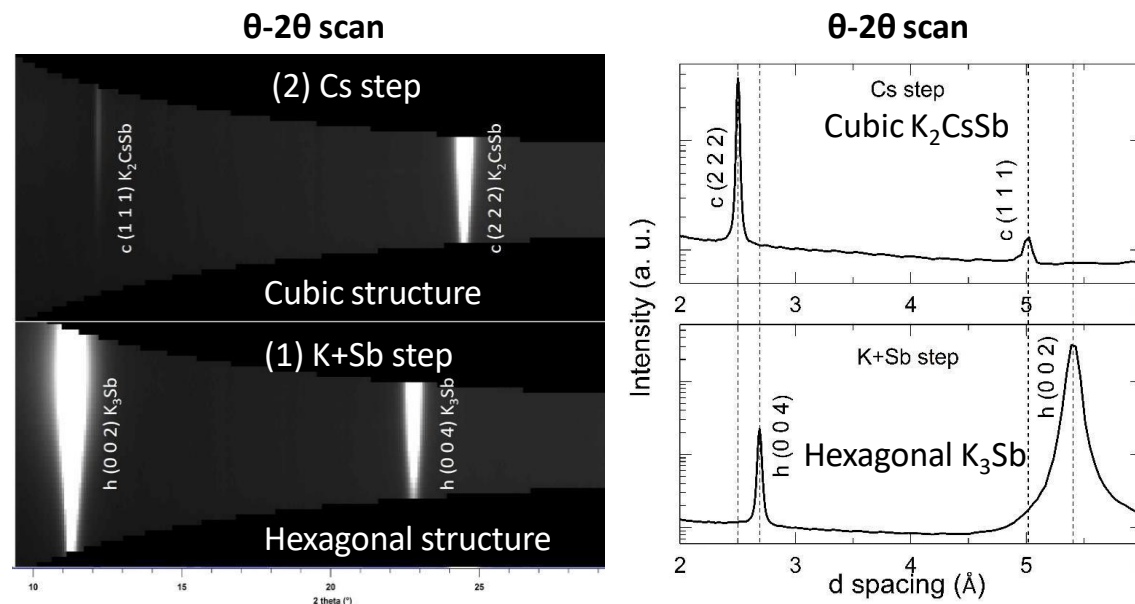
XRR

Sub-nm roughness  
 Similar thickness, roughness  
 from  $K_3Sb$  to  $K_2CsSb$



XRD

Full conversion from hexagonal  $K_3Sb$  to perfect  
 cubic  $K_2CsSb$   
 Textured XRD pattern

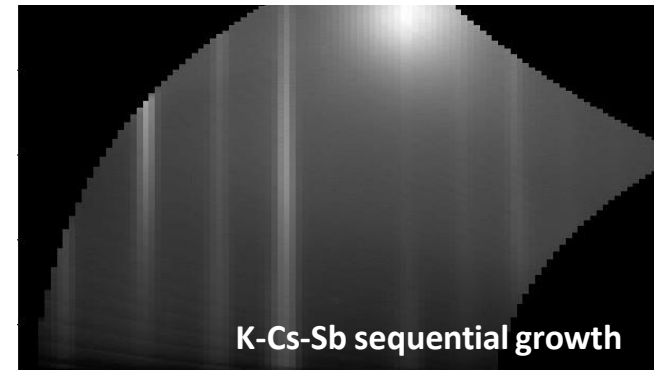
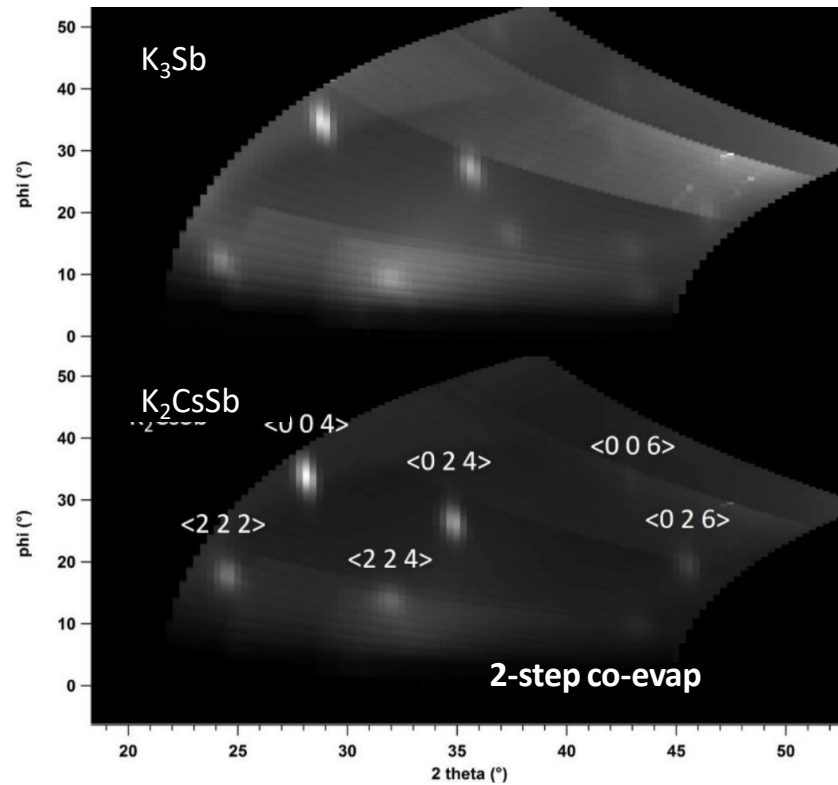


Diffraction arcs – textured film (both on Si (1 0 0) & Si (1 1 1))

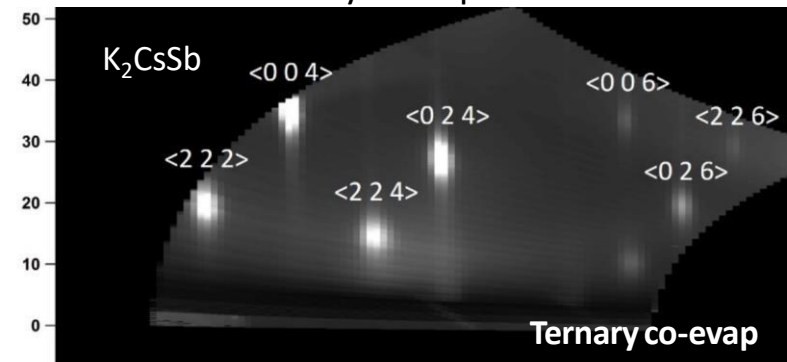
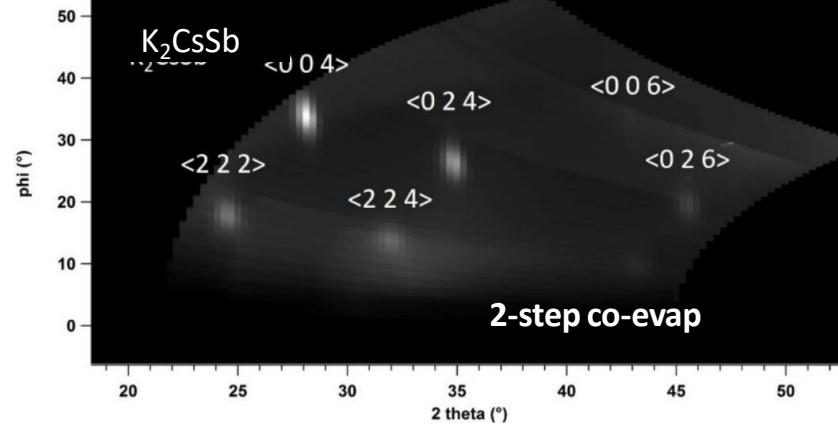
# Crystal Quality

Wide angle X-ray diffraction patterns

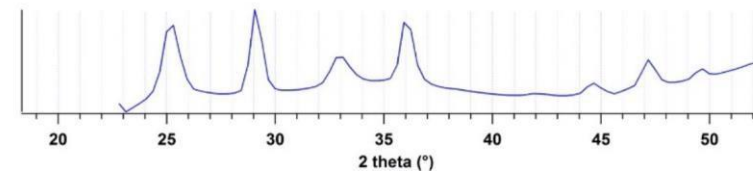
Two-step recipe



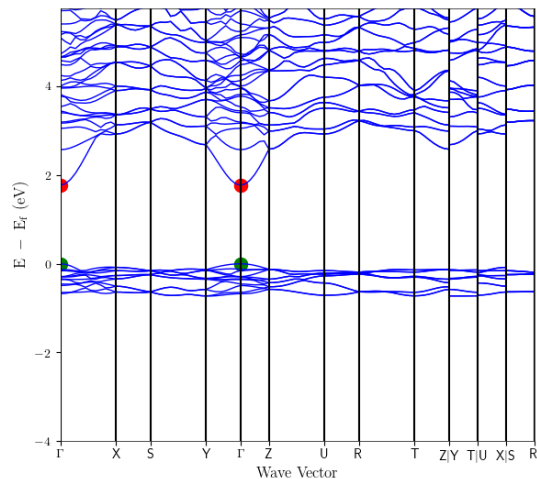
Ternary co-deposition



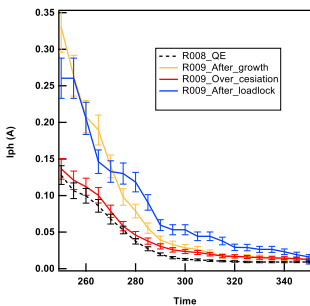
Large Grain, highly textured



# Example of success in one study being applied to adjacent problem: studies of $K_2CsSb$ enabled growth of $Cs_2Te$



HKL	Theory_d spacing	Exp_d spacing
222 $Cs_2Te$	2.307	2.315
111 $Cs_2Te$	4.613	4.615



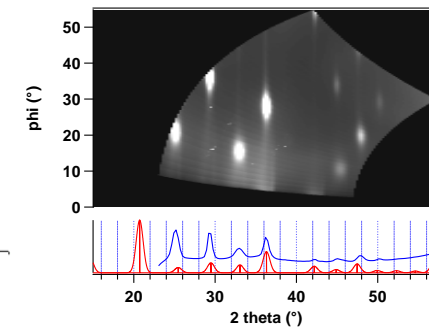
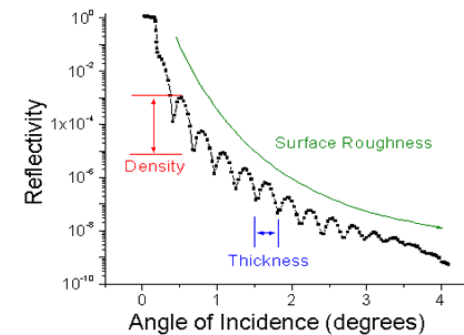
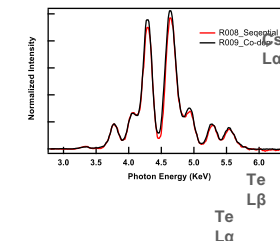
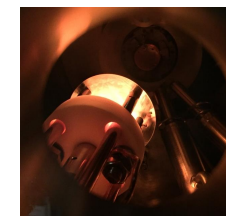
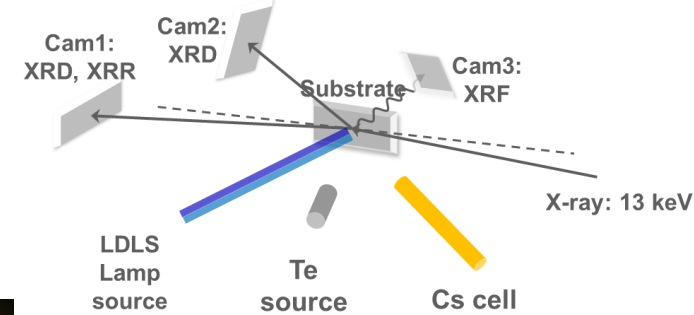
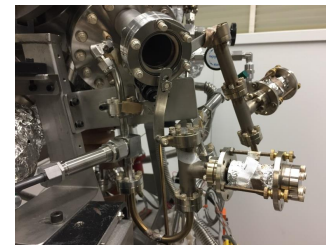
Predictive Theory & Validation

Advanced Nano-Material Synthesis

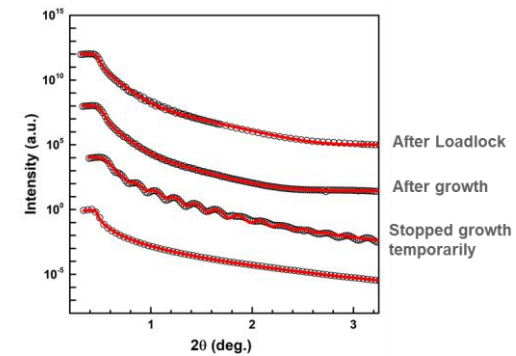
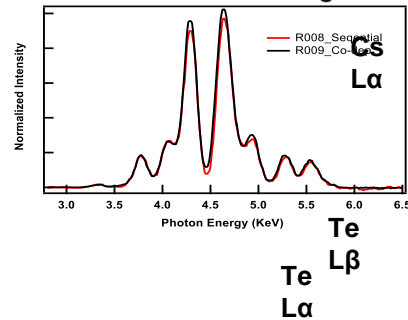
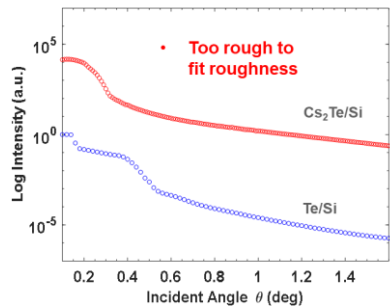
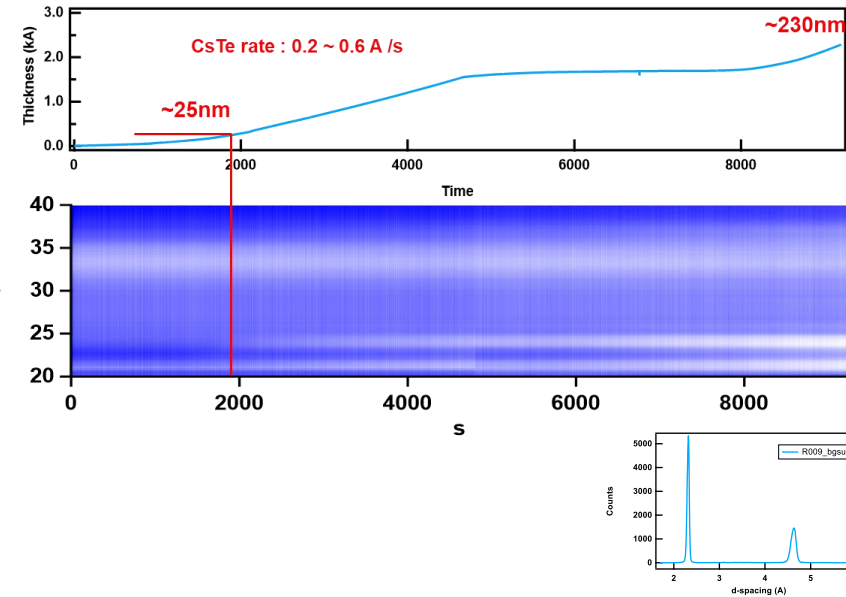
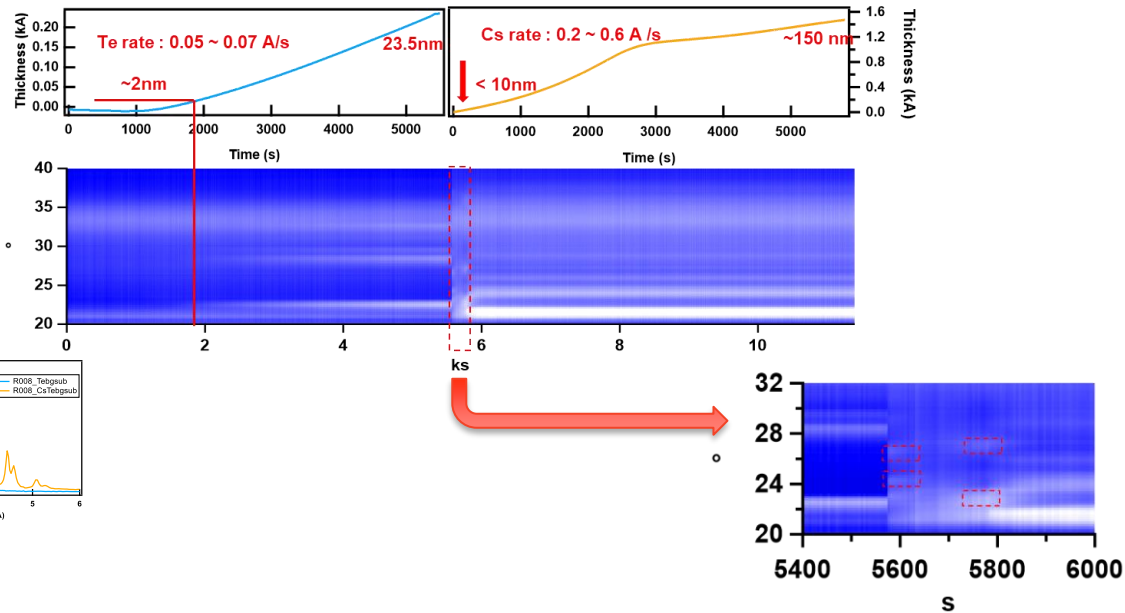
Correlated Emission Properties

X-ray characterization

Next step: transfer to procedure not requiring synchrotron light source



# Results presented by Mengjia Gaowei at P3(2018) show the efficacy of the materials by design approach: co-deposition gives smooth single phase

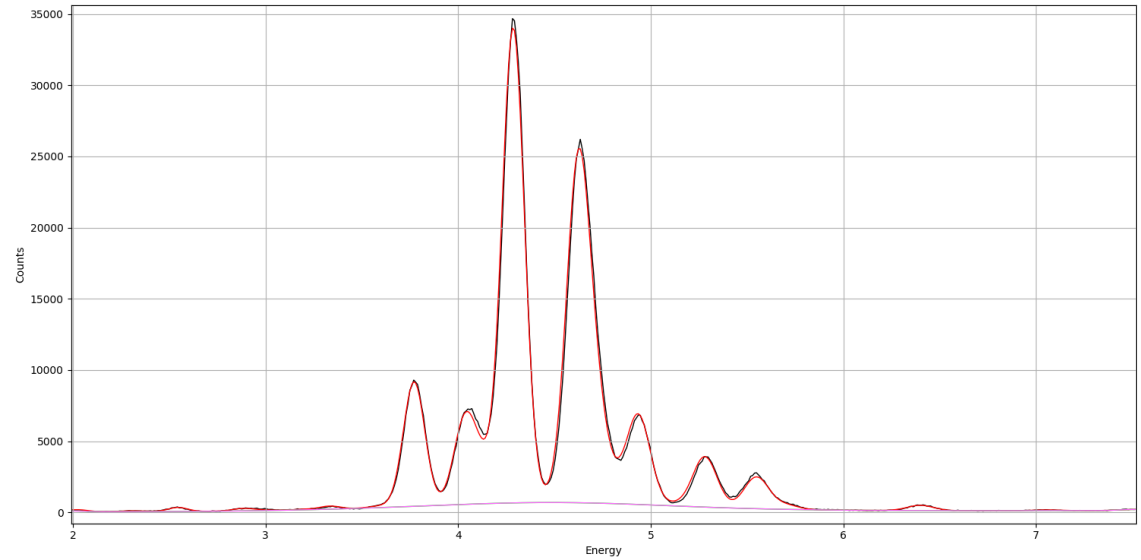
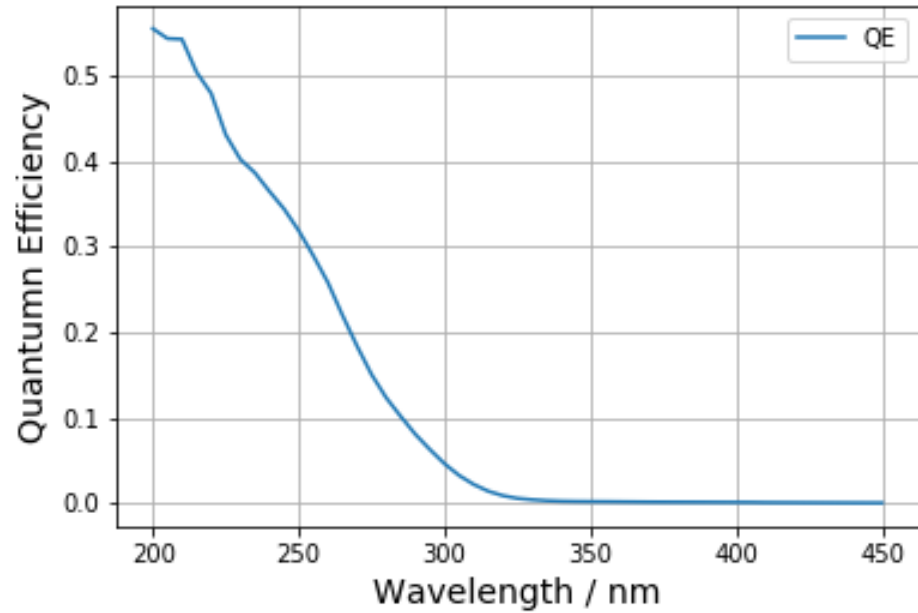


- Nearly all of the crystallized Te has dissolved
- Low counts in diffraction peaks
- Multiple phase of Cs-Te compound co-exist

- Well crystallized
- Single phase

# CsTe/Si: QE & XRF

CsTe	Te	Cs
Q007	1	1.98



M. Gaowei, J. Sinsheimer, D. Strom, J. Xie, J. Cen, J. Walsh, E. Muller, and J. Smedley  
Phys. Rev. Accel. Beams 22, 073401

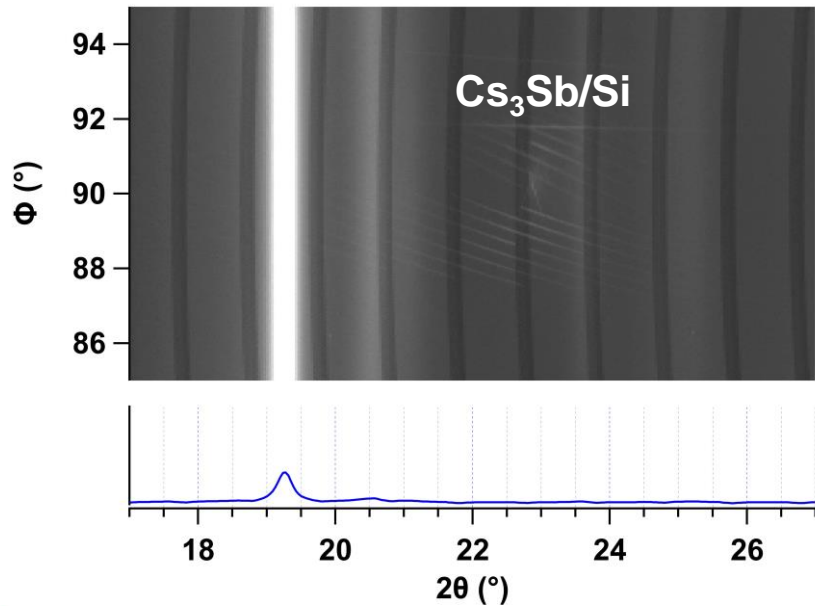
# Epitaxy

- **Epitaxy – alignment of crystals to a substrate, to create a pseudo-single crystal**
- **Gateway to complex materials engineering and heterostructuring**
- **X-rays are a good tool, but in this case electron diffraction is vital**

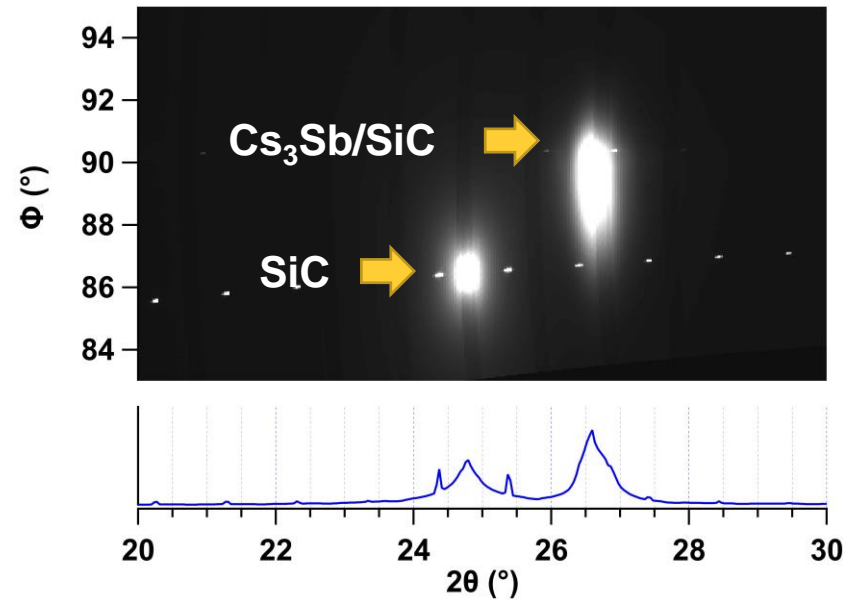


# Co deposition of $\text{Cs}_3\text{Sb}$ on 4-H SiC: Post growth Characterization

## X-ray Diffraction

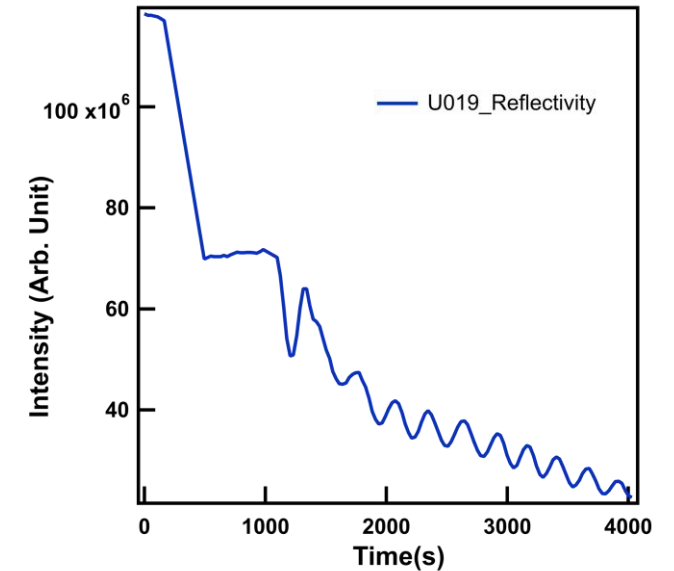


Film thickness 48.1 nm



Film thickness 20.5 nm

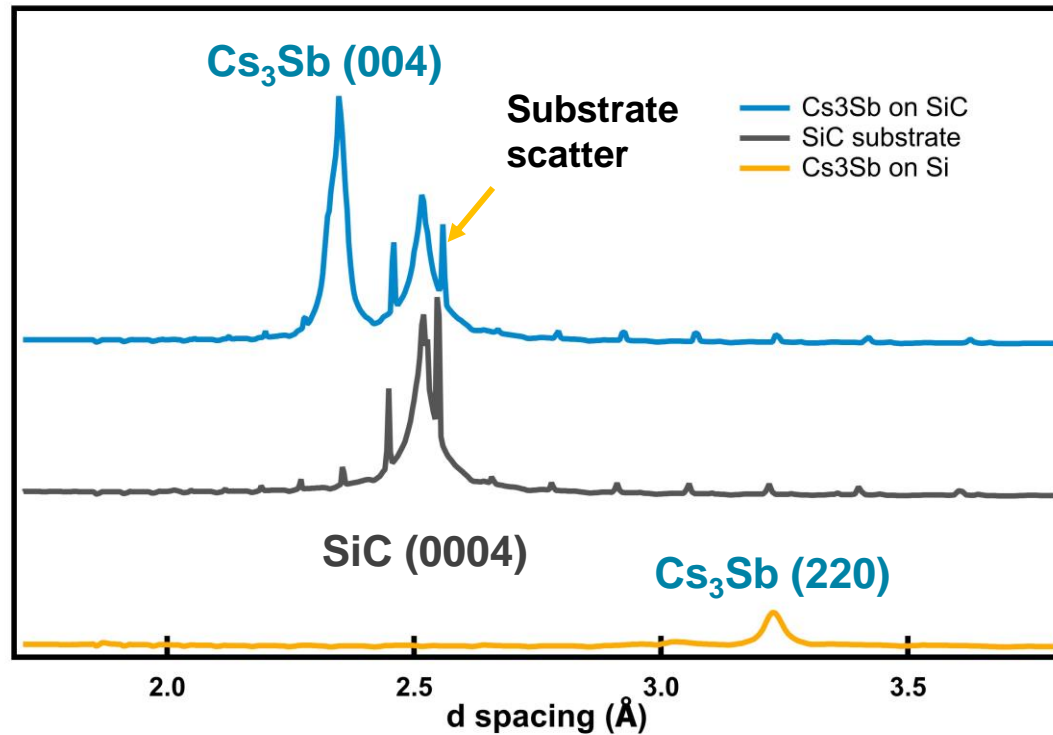
## X-ray reflectivity during growth



- Smooth film during growth
- Highly textured on SiC
- Possible epitaxy?

# Cs<sub>3</sub>Sb on 4-H SiC: Post growth Characterization

## X-ray Diffraction

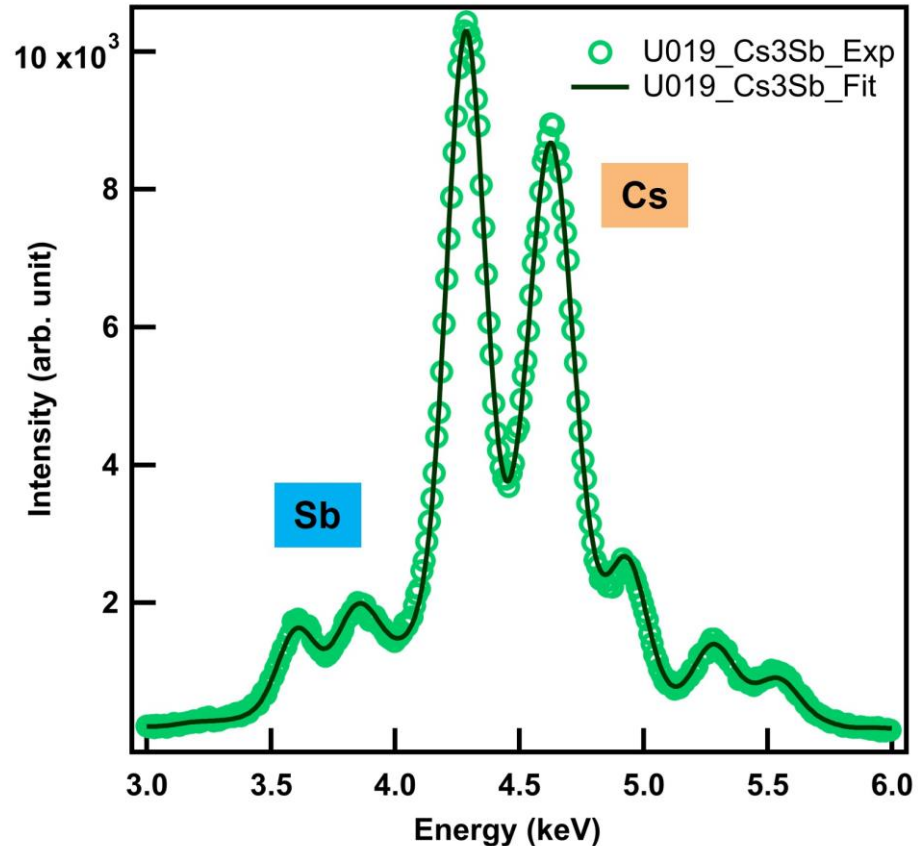


Diffraction peak	D spacing (Å)
Cs <sub>3</sub> Sb (004)	2.34
SiC (0004)	2.51
Cs <sub>3</sub> Sb (220)	3.23



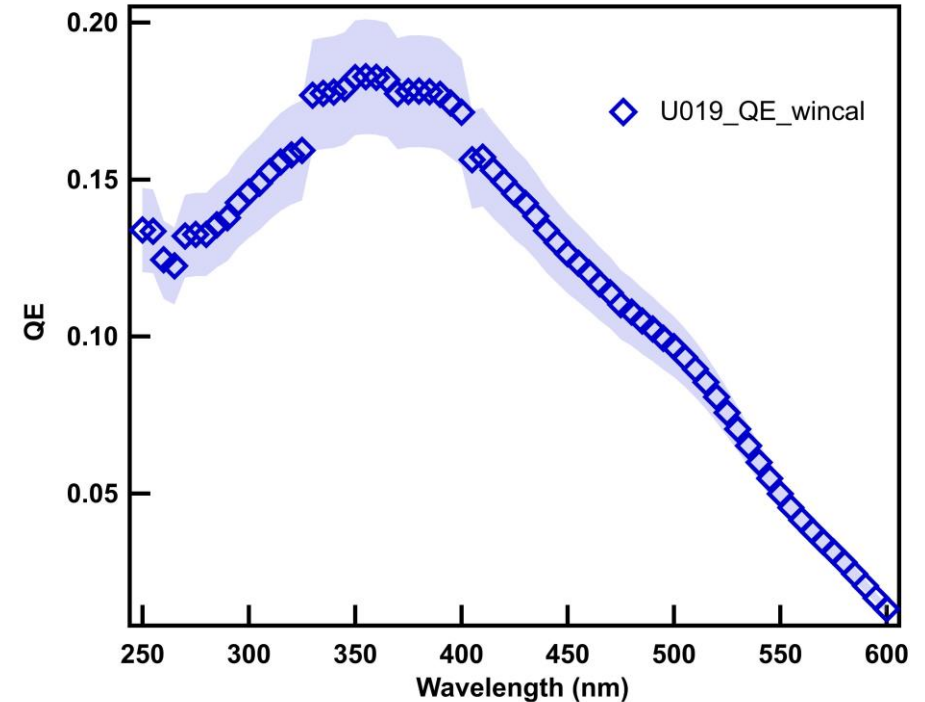
# Cs<sub>3</sub>Sb on 4-H SiC: Post growth Characterization

## X-ray Fluorescence



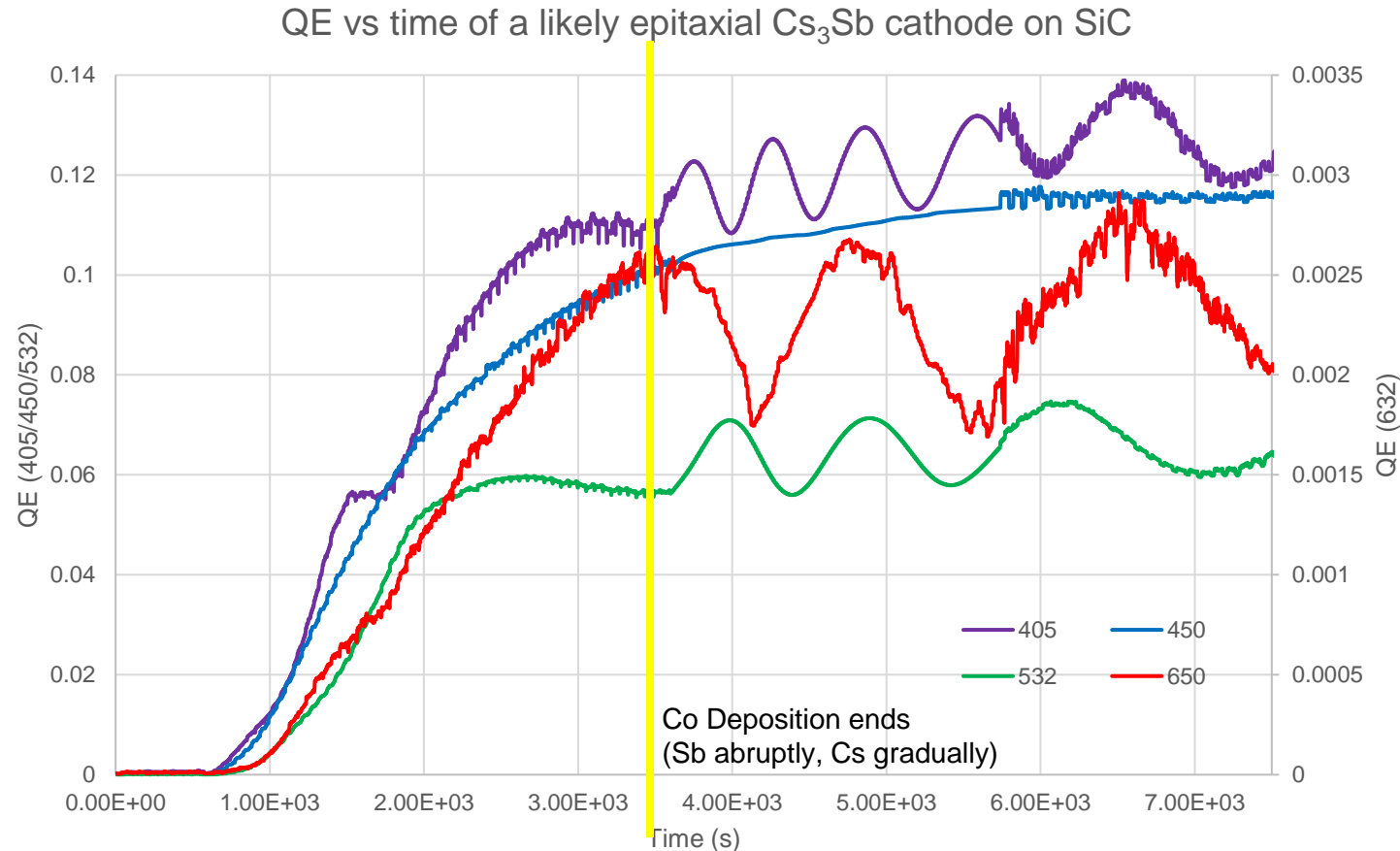
	K/Sb	Cs/Sb
Cs3Sb/4H-SiC	/	2.95

## Spectral Response



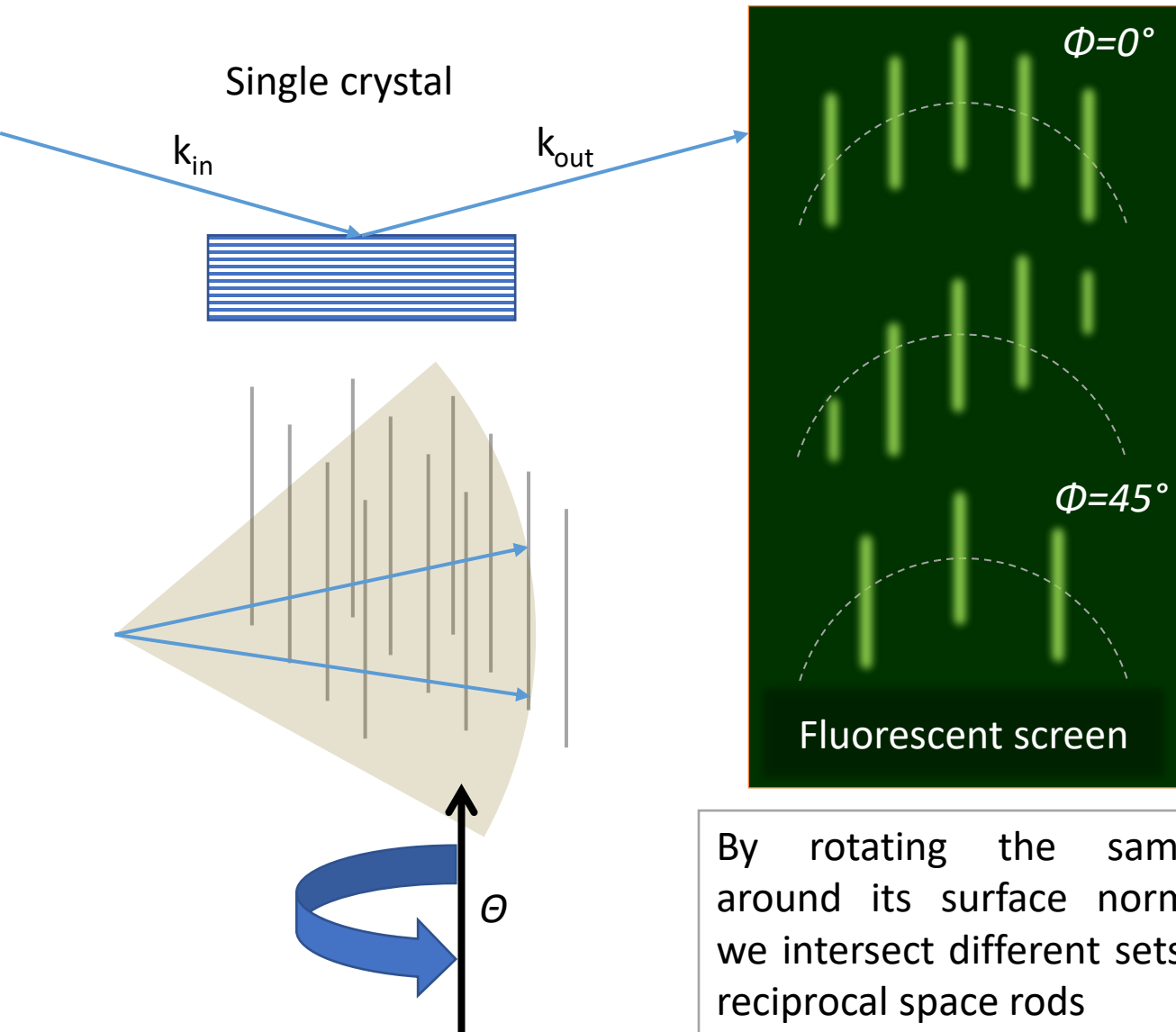
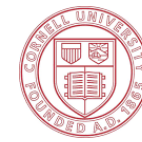
- 350 nm (peak): 18.2%
- 530 nm: 7%

# Cs<sub>3</sub>Sb on 4-H SiC: QE and optical etalon effect



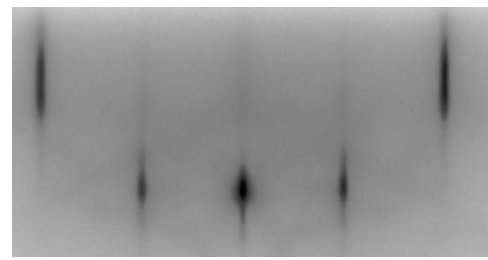
- Cathode is evolving after growth is stopped and substrate cooling down.
- Change of index of refraction
- Loss of material

# Information provided by RHEED

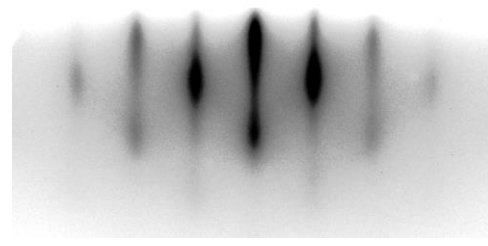


By rotating the sample around its surface normal, we intersect different sets of reciprocal space rods

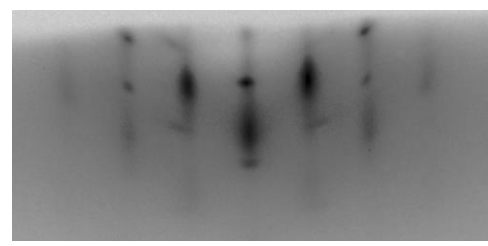
- Real-time
- Sub-ML sensitivity
- Qualitative probe of surface roughness and crystallinity



Single crystal  
High coherence

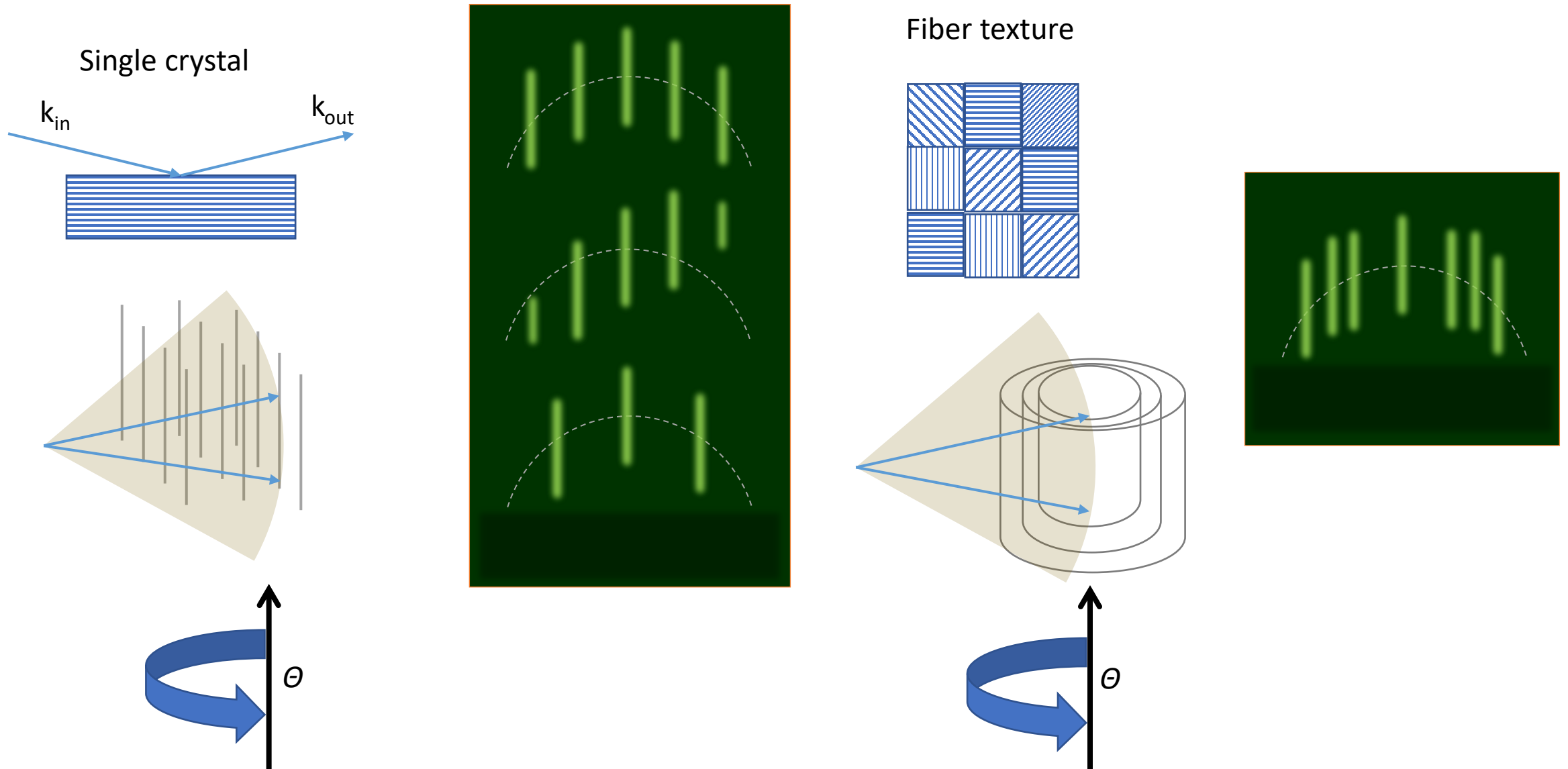


Film  
Reduced coherence  
Roughened surface



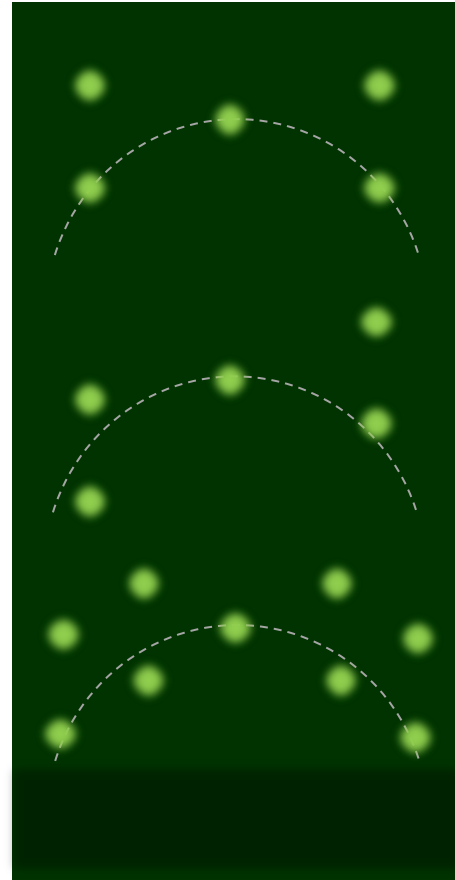
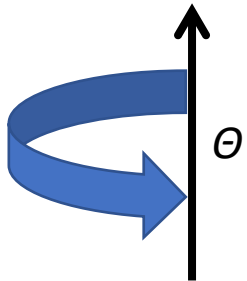
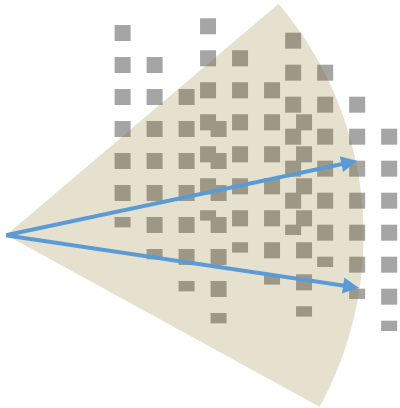
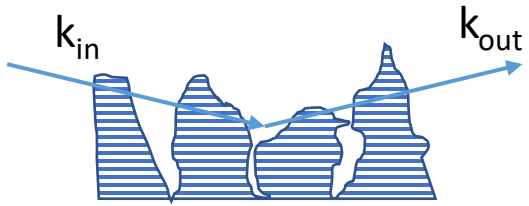
Film  
Polycrystalline domains/impurities

# Information provided by RHEED

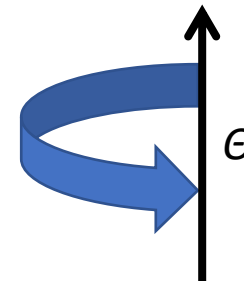
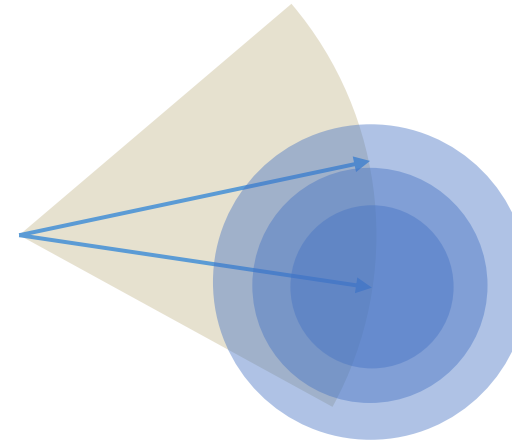
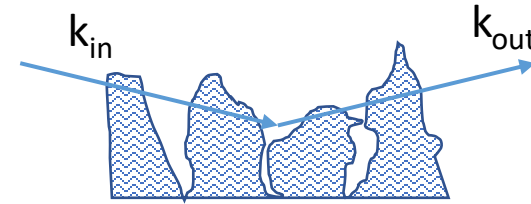


# Information provided by RHEED

Single crystal islands



Polycrystalline islands



No texture



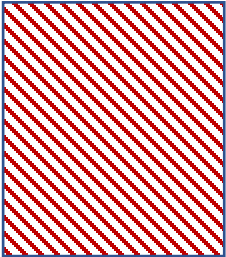
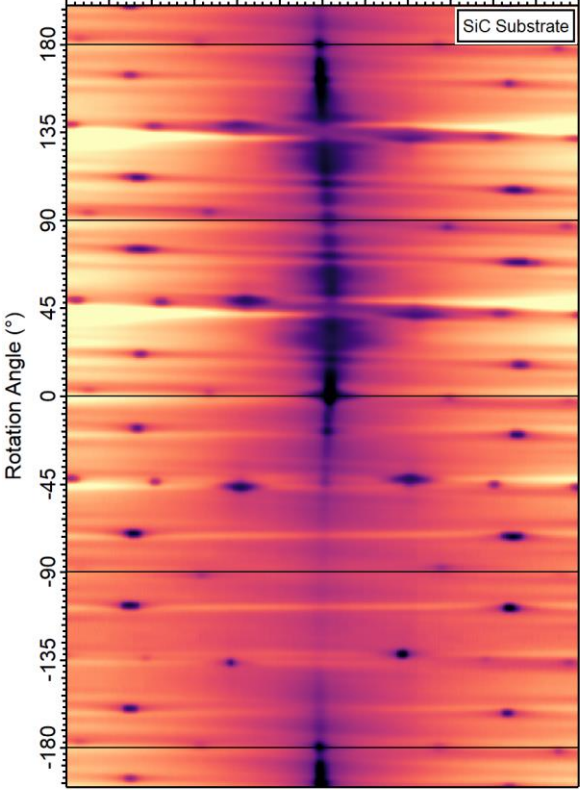
Textured film



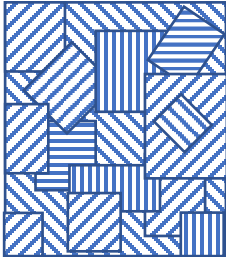
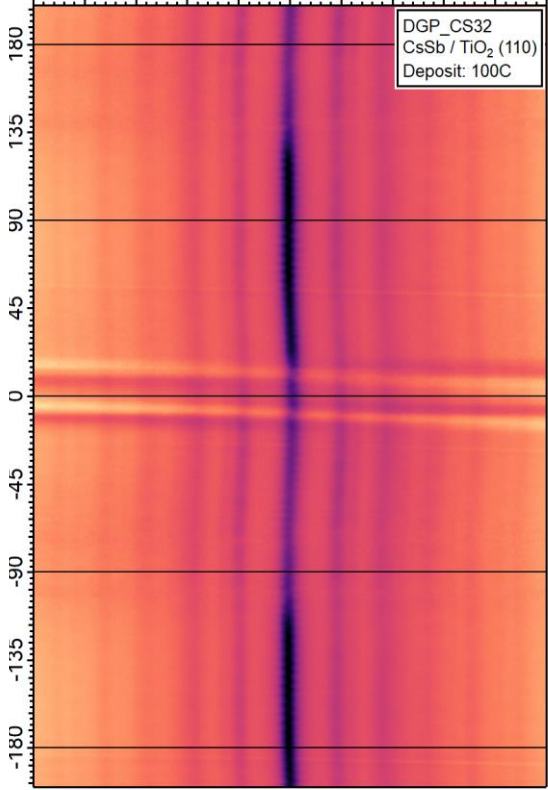
No rotation dependence if the texture axis is out-of-plane (uniaxial)  
Rotation dependence if the texture axis is in-plane (biaxial)

# Epitaxial Relationship

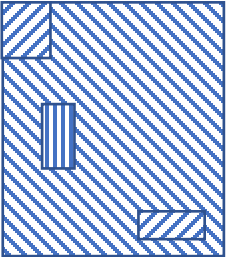
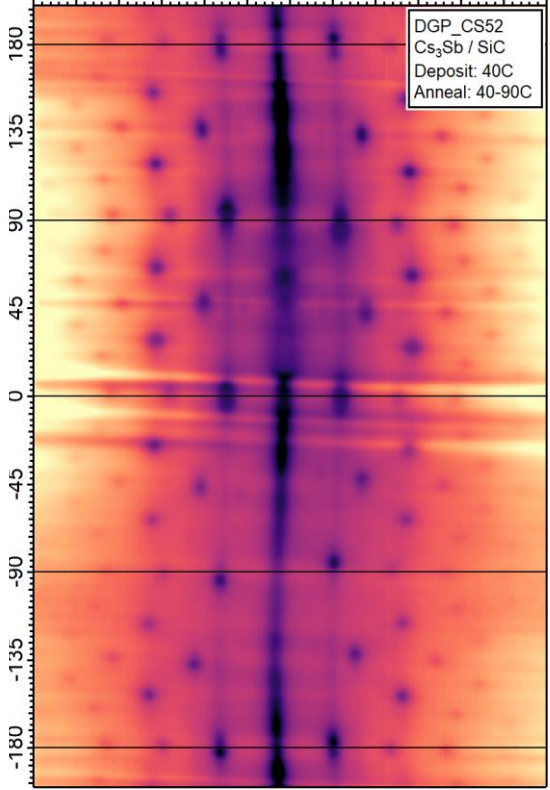
Substrate  
Fully Ordered



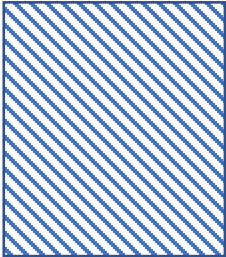
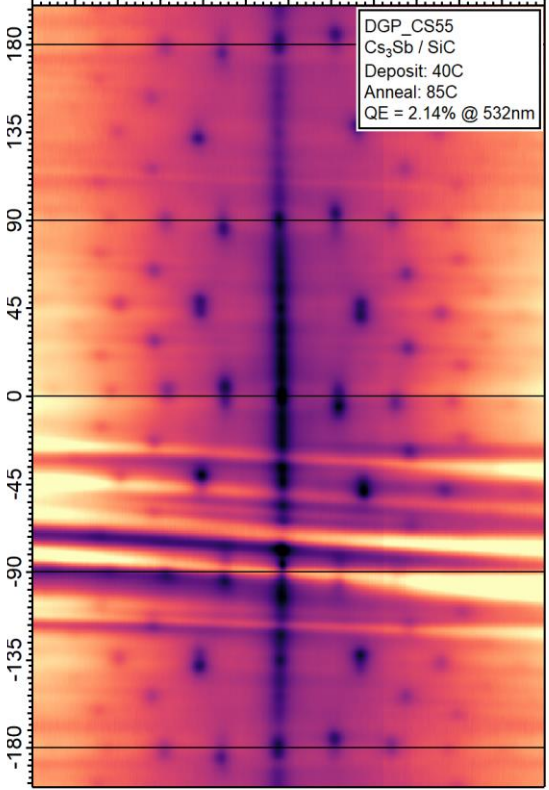
Fiber Textured Film  
Only c-axis oriented



Partially Ordered Film

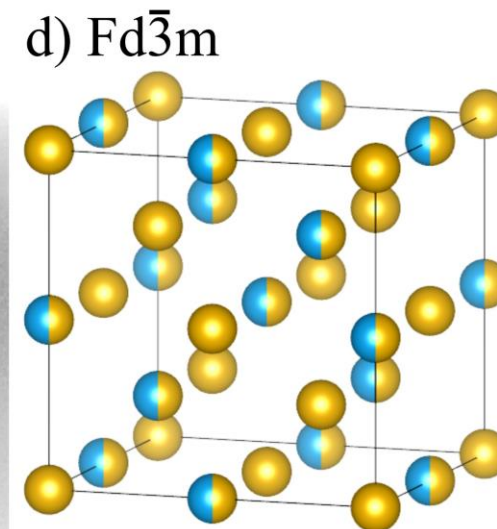
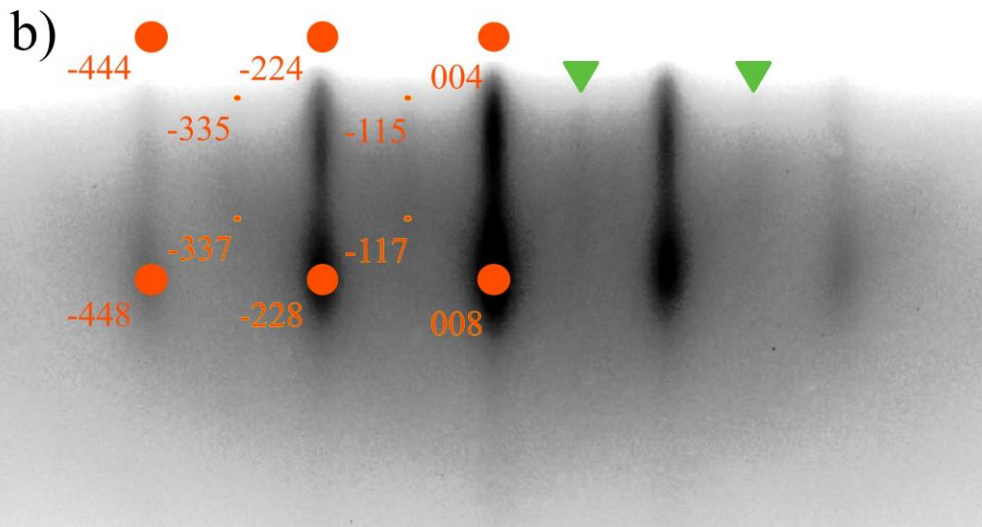
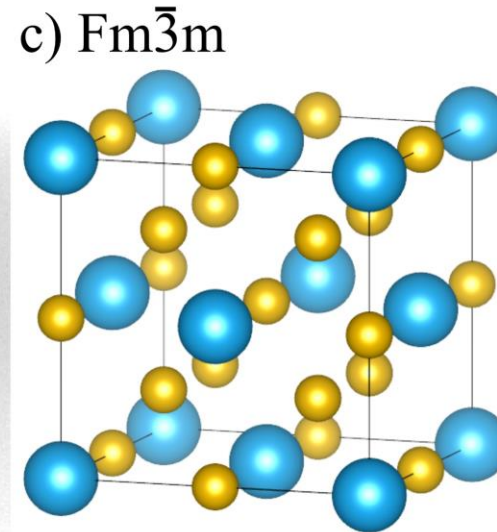
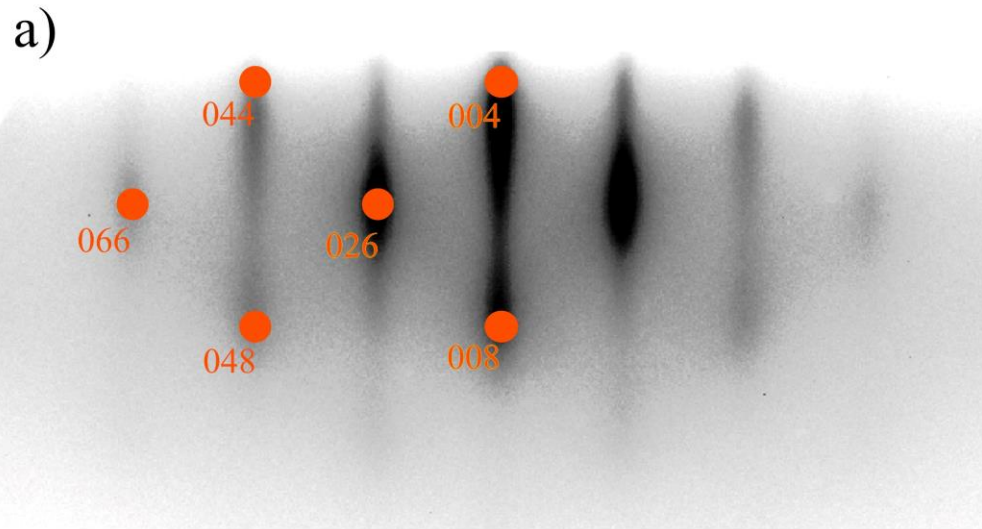


Epitaxial Film





# RHEED of epitaxial $\text{Cs}_3\text{Sb}$



To prove that indeed that the epitaxial relationship is  $[100]\text{SiC} // [100]\text{Cs}_3\text{Sb}$  I plot the reciprocal lattice of  $\text{Fd}\bar{3}\text{m}$   $\text{Cs}_3\text{Sb}$  superimposed to the RHEED patterns.

Half period streaks are predicted to appear in the  $[11]$  azimuth, as we observe.

Our data is compatible with any of the two literature reported structures.

# Final Thoughts

- The materials science community has developed a robust toolkit for optimizing thin films for various applications
- For materials where “dual uses” exist (GaAs), nearly perfect crystals, with heterostructuring, doping gradients and such exist
- For the PEA semiconductors, we still have work to do, but it is an exciting time. The dawn of epitaxy in these systems heralds a revolution in optimization.

## To-Do list:

Cross-correlate X-ray techniques with non-synchrotron measurements

Develop epitaxy for Ternary systems and  $\text{Cs}_2\text{Te}$

Demonstrate functional heterostructures

Measure conductivity vs temperature – better crystals = less resupply



# Emittance leverage for XFELs: Electron source sets ultimate limits on achievable electron beam quality, which sets ultimate limits on photon beam

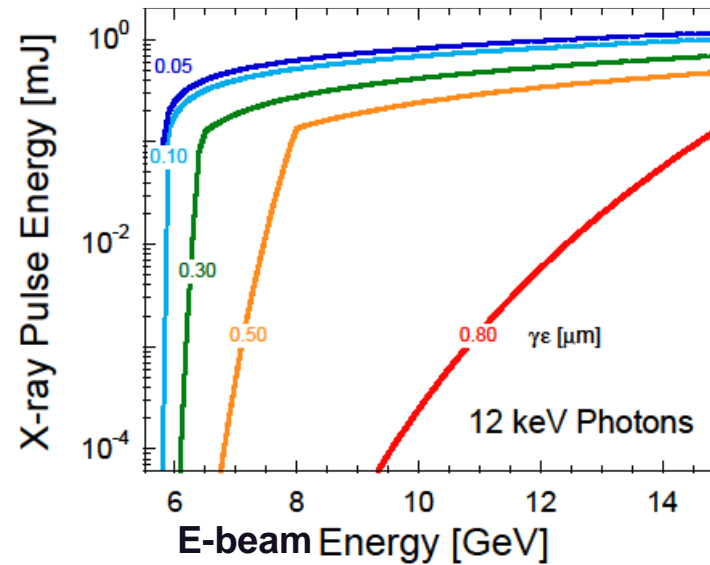
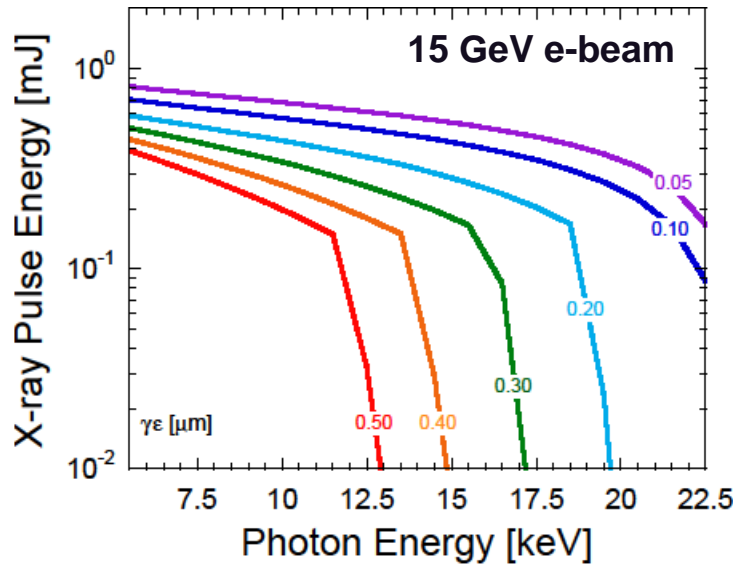
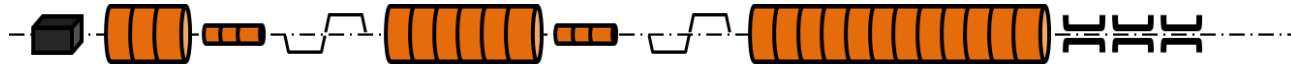


TABLE II. LCLS-II/HXR Case Study Parameters

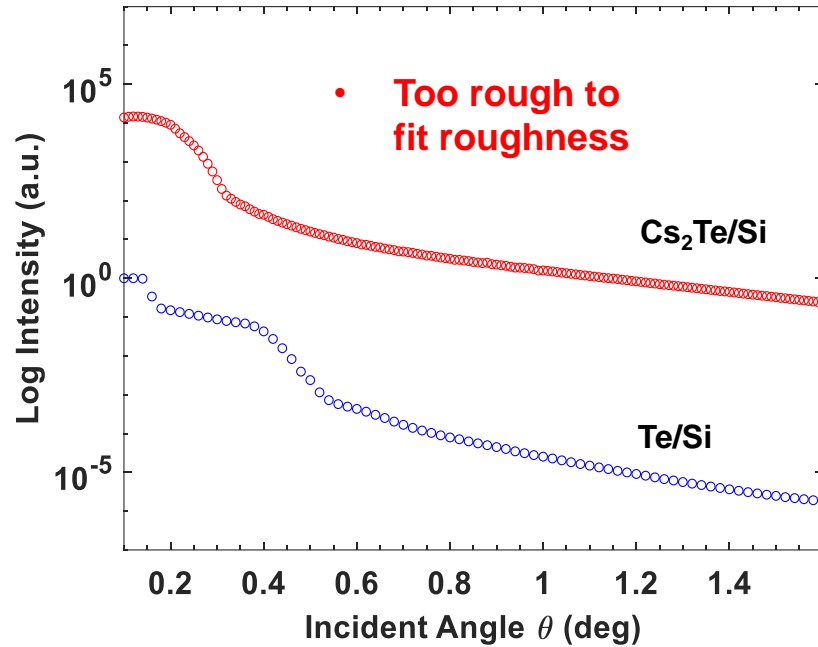
Parameter	Definition	Value
$E_b$	Beam Energy	15 GeV
$\sigma_\eta$	Energy Spread	1.5 MeV
$L_u$	Undulator Length	140m
$\lambda_u$	Undulator Period	26mm
$I_{peak}$	Peak Current	3.5 kA
$\beta$	Mean Beta	30 m
$Q_b$	Bunch Charge	100 pC

- *Photocathode upgrade presents a low-cost investment for improved performance of existing machines*
- *Emittance suppression at the cathode enables machine designs with lower emittance budget*

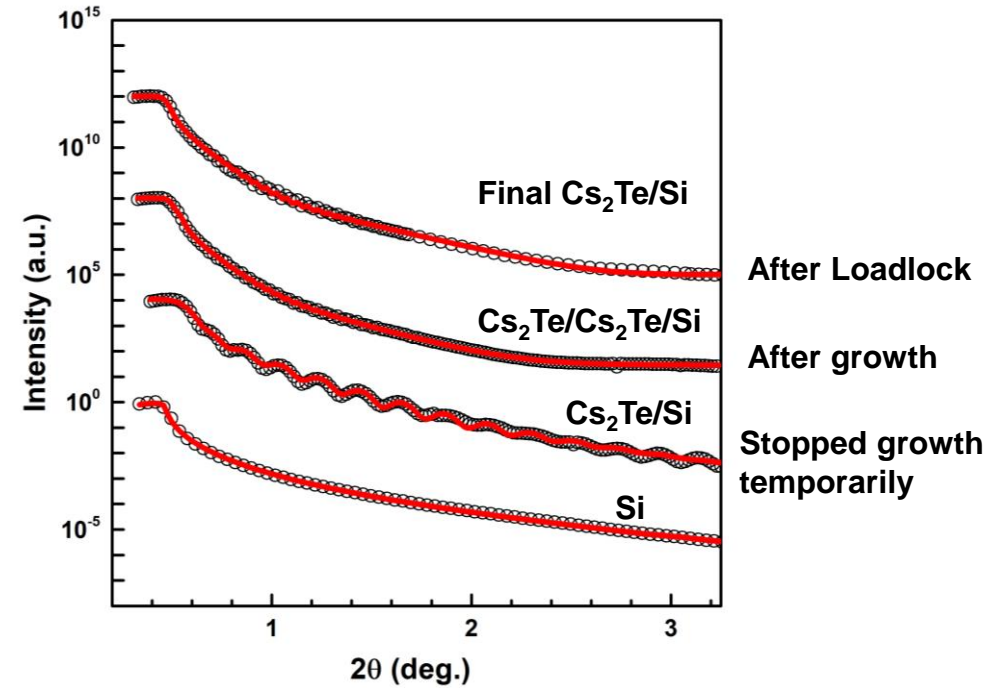
Moody, N.A., et al., *Perspectives on Designer Photocathodes for X-ray Free-Electron Lasers: Influencing Emission Properties with Heterostructures and Nanoengineered Electronic States*. Physical Review Applied, 2018. **10**(4): p. 047002.

# CsTe cathode surface roughness: XRR analysis

R008 Sequential - CsTe on Si



R009 Co-dep - CsTe on Si



	Thickness (Å)	Roughness (Å)
FINAL Cs <sub>2</sub> Te/Cs <sub>2</sub> Te	968.3 ± 2.9 (total Cs <sub>2</sub> Te)	19.1 ± 0.2
Cs <sub>2</sub> Te/Cs <sub>2</sub> Te	1026.1 ± 1.6 (total Cs <sub>2</sub> Te)	19.10 ± 0.07
Cs <sub>2</sub> Te	245.5 ± 1.7	9.55 ± 0.14
Si Substrate	-	3.75 ± 0.02

