#### FYS 4340/FYS 9340

## Diffraction Methods & Electron Microscopy

## Lecture 2

Sandeep Gorantla

UiO **Centre for Materials Science and Nanotechnology (SMN)** 



# Transmission Electron Microscopy

### Introduction and Basics Part- 1



Sandeep Gorantla

UiO **Centre for Materials Science and Nanotechnology (SMN)** 

# Learning more about TEM!

## David B. Williams • C. Barry Carter Transmission Electron Microscopy

A Textbook for Materials Science



Second Edition

Springer

Re	off Emil or the the sta
	E B A B
	668 \$
A	berration-Corrected
Ir	naging in Transmission
E	lectron Microscopy
	An Introduction 2nd Edition
	Imperial College Press

Courtesy: WWW.amazon.com



No.

**Optical Sciences** 

Ludwig Reimer

Electron

of Image Formation and Microanalysis

> Springer-Verlag Berlin Heidelberg GmbH

Physics

Third Edition

Transmission

Microscopy

# Learning more about TEM!



#### http://www.matter.org.uk/tem/

# Learning more about TEM!



# Why learn about Transmission Electron Microscopy (TEM)?





Search Powered by Google

Materials Science

Semiconductors

Life Sciences

Oil and Gas

Industrial Manufacturing Minerals and Mining



### Role of TEM in Materials Science Research and Development



Solving Materials Science problems/mysteries by probing analytically and understanding structure-property relationships at atomic scale level

Courtesy: www.wikipedia.com



#### **Allotropes of carbon**



(Courtesy: The Royal Swedish Academy of Sciences)



Courtesy: www.extremetech.com

AND COLORISE

The second

COLUMN STATE

のないないない

A CONTRACTOR OF THE OWNER OWNER

100



*Courtesy*: Knut Urban, Nature Materials 10, 165–166 (2011)

#### **1D** nanomaterials modification in TEM

- Irradiation of solids with energetic particles usually leads to damage
- However, in the case of carbon nanostructures, electron irradiation was observed to have some beneficial effects
  - (a) Irradiation mediated engineering(b) self-assembly or self-organization





Courtesy: Krasheninnikov, A. V. et al., Nature Mater., 6, 723 (2007)







#### Interface: defects on outer-wall of a nanotube and fullerene



Courtesy: Gorantla, S. et al., Nanoscale, 2, 2077 (2010)



### <u>Interface</u>: defects on outer-wall of a nanotube and fullerene

Nanohump formation (Covalent interactions of fullerene fusion)



#### Movie Settings:

•Frame speed: 0.6 s

•Total Frames: 48

#### **Experimental conditions:**

•Acquisition time: 1 s

•Time gap between individual frames: 1s - 30s

•Total time: 14 mins

Courtesy: Gorantla, S. et al., Nanoscale, 2, 2077 (2010)



### Interface: defects on the outer-wall of a SWCNT and fullerene

#### Fullerene fusion with a nanohump (Covalent interactions of fullerene fusion)



#### Movie Settings:

•Frame speed: 0.6 s

•Total Frames: 48

#### **Experimental conditions:**

•Acquisition time: 1 s

•Time gap between individual frames: 1 s

Courtesy: Gorantla, S. et al., Nanoscale, 2, 2077 (2010)



# HETEROSOLAR PROJECT The aim of the work



Develop new solar cell devices base on ZnO/Cu<sub>2</sub>O heterojunctions coupled with convetional Si based solar cells

Properties determined by the structures, faults and interfaces.



- \* Theoretical eficiency ~20 %
- \* Highest exp. eficiency 1-4 %



UiO University of Oslo Centre for Materials Science and Nanotechnology (SMN) he thin films and their interfaces.









# Transmission Electron Microscope Brief History



FYS 4340/9340 course – Autumn 2016

# **Brief History: The first electron microscope**



Ernst Ruska: Nobel Prize in physics 1986

- Knoll and Ruska, first TEM in 1931
- Idea and first images published in 1932
- By 1933 they had produced a TEM with two magnetic lenses which gave 12 000 times magnification.



Electron Microscope Deutsches Museum, 1933 model

## **Brief History: The state-of-art TEM**



Electron Microscope Deutsches Museum, 1933 model



FEI Titan 60-300 TEM, NORTEM facility- UiO Installed: 2014

## **Brief History: The state-of-art TEM**

**BIG LEAP**: Introduction of **Lens Aberration Correctors** allowing atomic resolution at low accelerating voltages.

# **Resolution limit**

Year	Resolution			
1940s		~10nm		
1950s	~0.5-2nm			
1960s	0.3nm (transmission)			
17008	~15-20nm (scanning)			
$1070_{S}$	0.2nm (transmission)			
19708	7nm (standard scanning)			
1090	0.15nm (transmission)			
19808	5nm (scanning at 1kV)			
1000	0.1nm (transmission)			
19908	3nm (scanning at 1kV)			
2000s		<0.1 nn	(Cs correctors)	

Typical TEM operating voltages in Materials Science Research





Core of the M100 galaxy seen through Hubble (source: NASA)

Courtesy: http://www.sfc.fr/Material/hrst.mit.edu/hrs/materials/public/ElecMicr.htm



# Transmission Electron Microscope Fundamentals



## Electrons interaction with the specimen





## **Electron lenses**

Any axially symmetrical electric or magnetic field have the properties of an ideal lens for paraxial rays of charged particles.

- Electrostatic **F=-eE** 
  - Not used as imaging lenses, but are used in modern monochromators

- ElectroMagnetic F= -e(v x B)
  - Can be made more accurately
  - Shorter focal length



Courtesy: http://www.matter.org.uk/tem/lenses/electromagnetic\_lenses.htm



## **TEM Lens Aberrations**







## **TEM Lens Aberrations**



Courtesy: Knut W. Urban, Science 321, 506, 2008; CEOS gmbh, Germany; www.globalsino.com



## **TEM Lens Aberrations**





# Transmission Electron Microscope Instrumentation – Part 1





UiO **University of Oslo** Centre for Materials Science and Nanotechnology (SMN)











#### Specimen Stage





## **TEM Specimen Holder**





### **TEM Specimens**



• Typically 3 mm in diameter

Courtesy: http://asummerinscience.blogspot.no





#### **TEM Viewing Chamber – Phosphorous Screen**

MPANY-





#### TEM Image recording CCDs and EELS Spectrometer



# Transmission Electron Microscopy

Introduction and Basics Part-2



UiO **U**IO **University of Oslo** Centre for Materials Science and Nanotechnology (SMN)

### **TEM in Materials Science**

The interesting objects for TEM is not the average structure or homogenous materials but local structure and inhomogeneities





## **TEM techniques**

#### Main Constrast phenomena in TEM

### **Imaging**

Conventional TEM Bright/Dark-Field TEM High Resolution TEM (HRTEM) Scanning TEM (STEM) Energy Filtered TEM (EFTEM)

## **Diffraction**

Selected Area Electron Diffraction Convergent Beam Electron Diffraction

## **Spectroscopy**

Electron Dispersive X-ray Spectroscopy (EDS) Electron Energy Loss Spectroscopy (EELS) •Mass thickness Contrast
•Diffraction contrast
•Phase Contrast
•Z-contrast

Phase identification, defects, orientation relationship between different phases, nature of crystal structure (amorphous, polycrystalline, single crystal)

Chemical composition, electronic states, nature of chemical bonding (EDS and EELS). Spatial and energy resolution down to the atomic level and ~0.1 eV.



### **Objective aperture: Contrast enhancement**



All electrons contributes to the image.

Intensity: Thickness and density dependence

#### Mass-thickness contrast



A small aperture allows only electrons in the central beam in the back focal plane to contribute to the image.



Diffraction contrast (Amplitude contrast)

One grain seen along a low index zone axis.



## **TEM techniques**

#### Simplified ray diagram of conventional TEM

## Imaging

Conventional TEM Bright/Dark-Field TEM High Resolution TEM (HRTEM) Scanning TEM (STEM) Energy Filtered TEM (EFTEM)

## **Diffraction**

#### Selected Area Electron Diffraction Convergent Beam Electron Diffraction

## **Spectroscopy**

#### Electron Dispersive Spectroscopy (EDS) Electron Energy Loss Spectroscopy (EELS)





## Imaging



Courtesy: http://www.ifam.fraunhofer.de; I.MacLauren et al, International Materials Review, 59, 115 (2004)



## Imaging

TEM

STEM



Mass thickness and diffraction contrast

Mass thickness and Z- contrast



## Imaging

HRTEM



Z- contrast



Phase contrast

47 FYS 4340/9340 course – Autumn 2016

#### HAADF-STEM

HRTEM



Raw HAADF-STEM, ABF-STEM and HRTEM image of Si in the [110] zone axis by FEI Titan 60-300 with spatial resolutions of 0.8 Å for STEM and 2.0 Å for TEM.

Courtesy: Wei Zhan, Øystein Prytz, et al. (2015), SMN, UiO



## **Electron Diffraction in TEM**



#### Simplified ray diagram





## **Electron Diffraction in TEM**

#### **Elastic scattered electrons**

Only the direction of  $\mathbf{v}$  is changing. (Bragg scattering)

Elastic scattering is due to Coulomb interaction between the incident electrons and the electric charge of the electron clouds and the nucleus. (Rutherford scattering).

The elastic scattering is due to the average position of the atoms in the lattice.

Reflections satisfying Braggs law:

#### 2dsinθ=nλ



#### **Electrons interacts 100-1000 times stronger with matter than X-rays**

-more absorption (need thin samples)
-can detect weak reflections not observed with XRD technique

Courtesy: Dr. Jürgen Thomas, IFW-Dresden, Germany



## Selected area diffraction(SAD)

- Parallel incoming electron beam and a selection aperture in the image plane.
- Diffraction from a single crystal in a polycrystalline sample if the SAD aperture is small enough/crystal large enough.
- Orientation relationships between grains or different phases can be determined.
- ~2% accuracy of lattice parameters
  - Convergent electron beam better





## **Camera constant**





## **Indexing diffraction patterns**

The **g** vector to a reflection is normal to the corresponding (h k l) plane and  $IgI=1/d_{nh nk nl}$ 



- Measure R<sub>i</sub> and the angles between the reflections
- Calculate  $d_i$ , i=1,2,3 (=K/R<sub>i</sub>)
- Compare with tabulated/theoretical calculated d-values of possible phases
- Compare R<sub>i</sub>/R<sub>j</sub> with tabulated values for cubic structure.
- $\mathbf{g}_{1,hkl} + \mathbf{g}_{2,hkl} = \mathbf{g}_{3,hkl}$  (vector sum must be ok)
- Perpendicular vectors:  $\mathbf{g}_i \bullet \mathbf{g}_j = 0$
- Zone axis: **g**<sub>i</sub> x **g**<sub>j</sub> =[HKL]<sub>z</sub>
  - All indexed **g** must satisfy:  $\mathbf{g} \bullet [HKL]_z=0$

## **Electron Diffraction in TEM**

#### Amorphous phase

#### Poly crystalline sample

Single Crystals Interface between two different phases epitaxially grown







The orientation relationship between the phases can be determined with ED.







## **Spectroscopy**



## **X-ray Energy Dispersive Spectroscopy**



#### We detect the X-rays generated by the sample on a spectrometer Each element has a unique atomic structure and hence a characteristic X-ray energy

![](_page_57_Picture_3.jpeg)

## **Energy Dispersive X-ray Spectroscopy**

![](_page_58_Picture_1.jpeg)

## **Electron Energy Loss Spectroscopy (EELS)**

![](_page_59_Figure_1.jpeg)

Courtesy: William & Carter, Transmission Electron Microscopy; EM group, Univ. of Nevada, Reno.

![](_page_59_Picture_3.jpeg)

# **Electron Energy Loss Spectroscopy (EELS)**

### EELS of the Oxygen K edge

![](_page_60_Figure_2.jpeg)

The reference spectra of  $Cu_2O$  and CuO are from online EELS database<sup>1</sup>. The reference spectra were shifted in energy to match the first O K peak in our experimental, and scaled by the total counts in the energy-loss 560-590 eV.

![](_page_60_Picture_4.jpeg)

<sup>1</sup>Ngantcha, Gerland, Kihn & Riviere, *Eur. Phys. J. Appl. Phys.* **29**, (2005) 83.

![](_page_60_Picture_6.jpeg)

![](_page_60_Picture_7.jpeg)

FYS 4340/9340 course – Autumn 2016 61

Next Lecture

## • TEM Instrumentation – Part 2 (Text book Chapters: 5 – 9)

## • TEM Specimen Preparation (Text book Chapters: 10)

![](_page_61_Picture_3.jpeg)