

Hydrogen Induced Stress Cracking (HISC) in duplex stainless steel caused by cathodic protection

9 October 2013

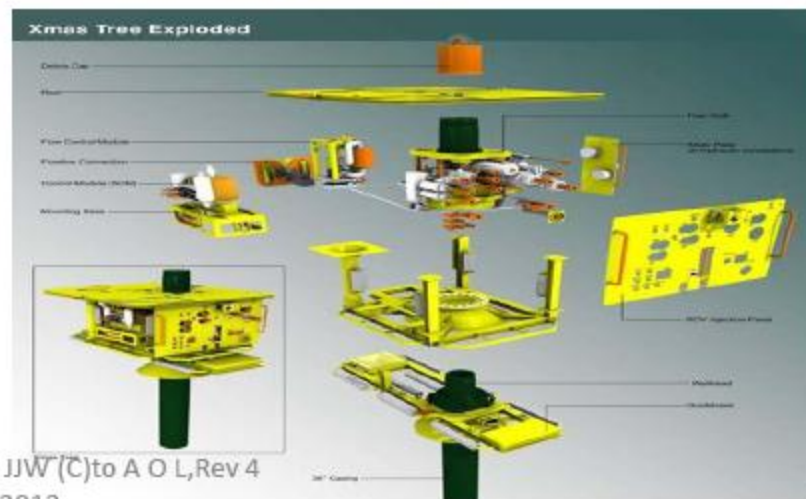
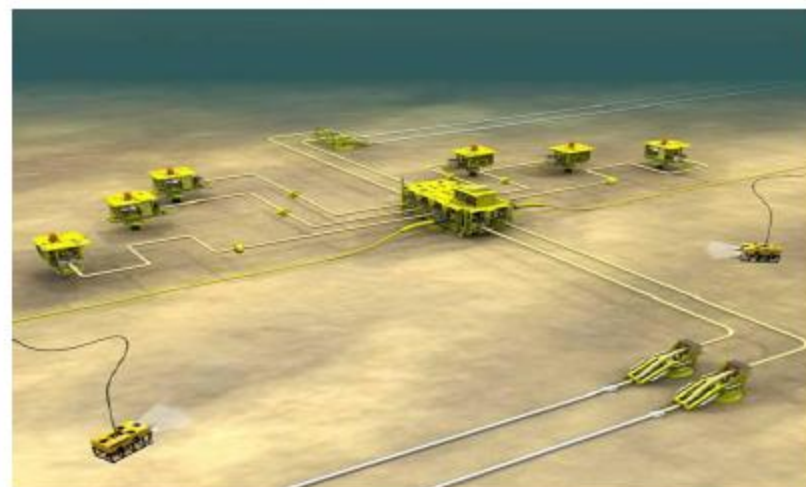
FTUI Metalurgi dan material

Kukuh W. Soerowidjojo
Metalurgi 80

Deepwater field



Total Angola - Dalia Field lay-out + SPS / SURF



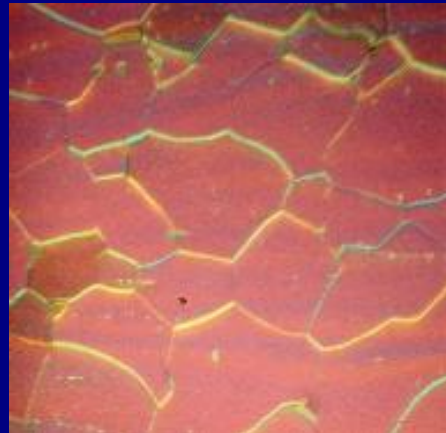
What is Duplex Stainless

- An equal percentage of ferrite and austenite is essential for the microstructure to the welded joint in SAF2507
- It is essential for UNS S32750 weld chemistry and HAZ to have a correct balance of ferrite and austenite optimum similar to the parent metal.
 - Austenite produces a toughness in the alloy
 - Ferrite helps provide rigidity and strength
 - UNS S32750 is alloyed so that both phases have the same corrosion resistance
 - By combining the best properties of both phases the alloy UNS S32750 develops excellent stress corrosion resistance essential for this service

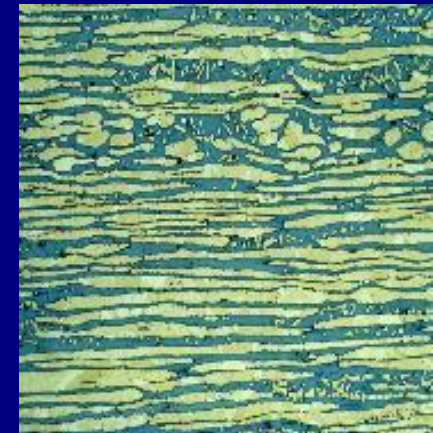
Austenite



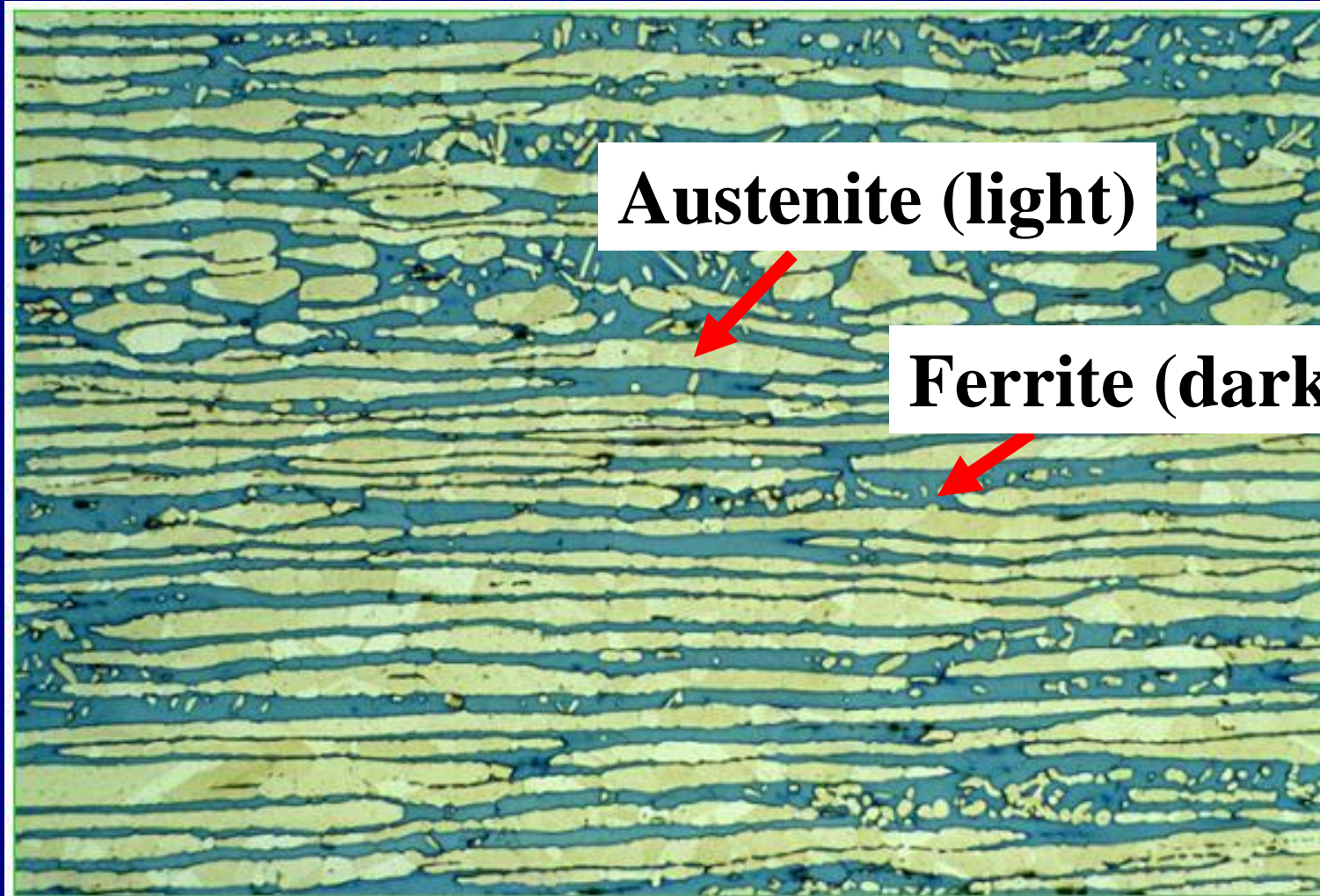
Ferrite



Duplex



Duplex Microstructure



Austenite (light)

Ferrite (dark)

Duplex Stainless Family

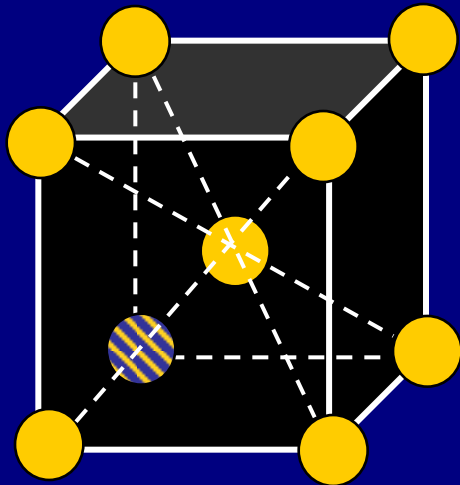


Alloy	C max	Si max	Mn max	P max	S max	Cr	Ni	Mo	Others
UNS S32101	0.030	1.0	5.0	0.04	0.03	21.5	1.5	0.3	PREN > 22
UNS S32304	0.030	1.0	2.0	0.035	0.015	22.5	4.5	0.3	Cu=0.3 N=0.1 PREN > 24
UNS S32205	0.030	1.0	2.0	0.030	0.015	22	5	3.2	N = 0.18 PREN > 35
UNS S32750	0.030	0.8	1.2	0.035	0.015	25	7	4	N = 0.3 PREN > 42
UNS S33207	0.030	0.8	1.5	0.035	0.010	32	7	3.5	N = 0.5 PREN > 50
UNS S32707	0.030	0.5	1.5	0.035	0.010	27	6.5	4.8	N=0.4 Co=1.0 PREN > 48

$$\text{PREN} = \% \text{ Cr} + 3.3 \times (\% \text{ Mo} + 0.5 \times \% \text{ W}) + 16 \times \% \text{ N}$$

Crystal Structures

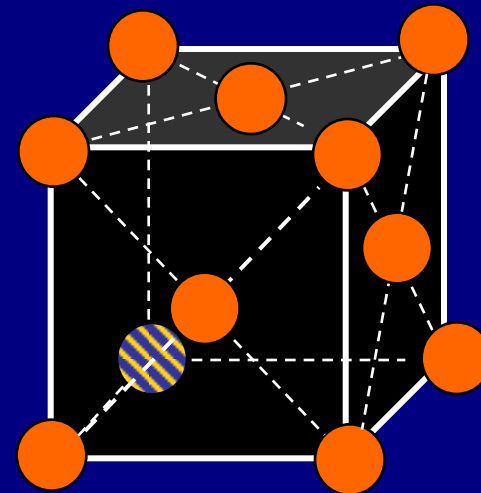
Ferrite



Body Centered
Cubic

Add
Nickel
→

Austenite



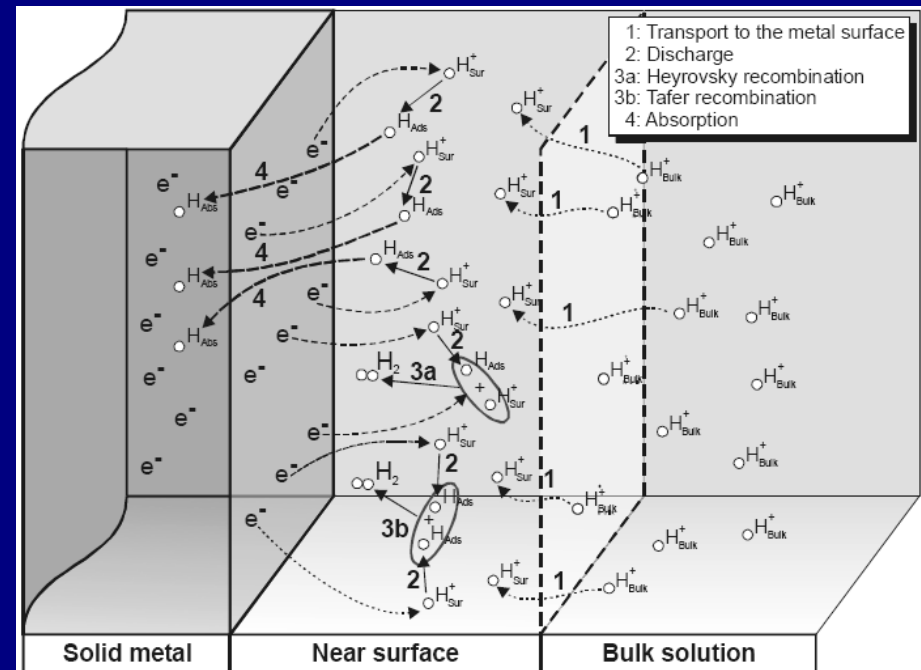
Face Centered
Cubic

Hydrogen formation and evolution

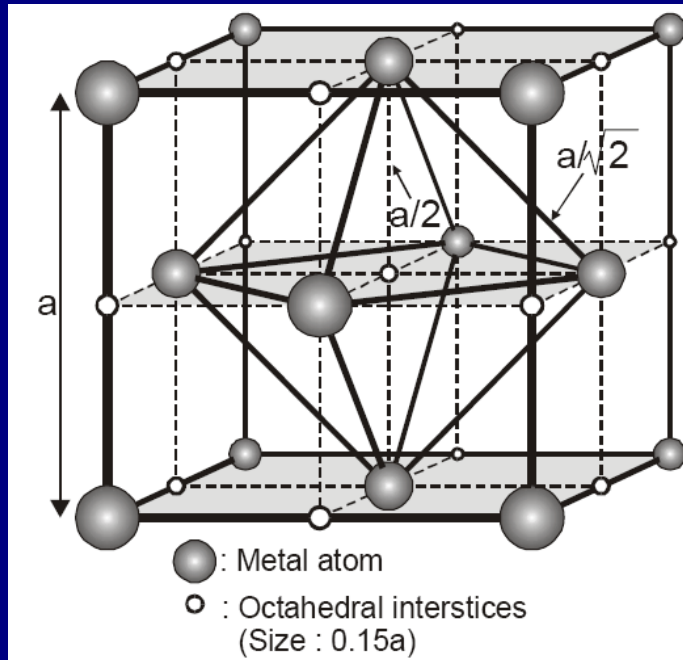
Cathodic reactions:



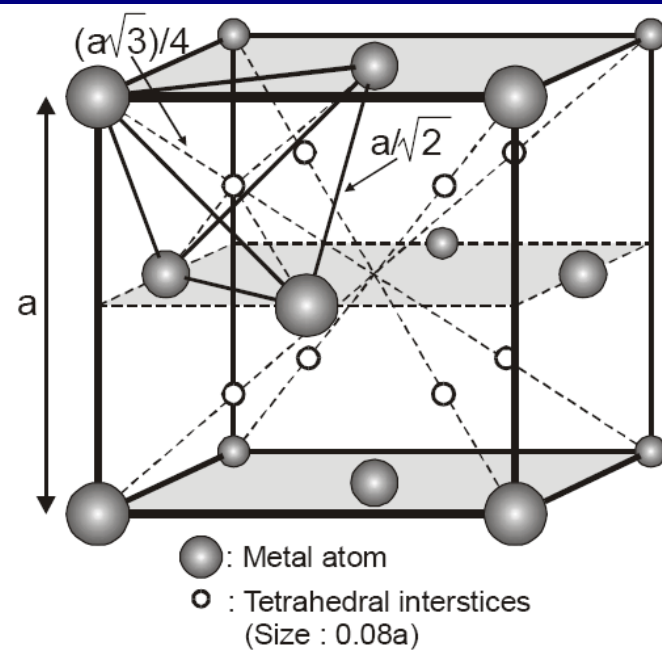
Hydrogen evolution



Hydrogen interstitial in austenite

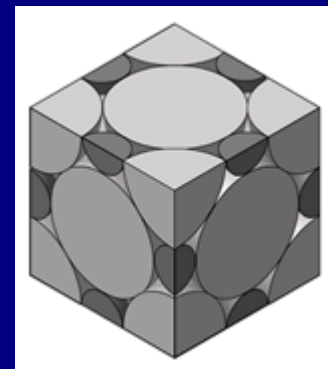


a) γ -Iron system (Octahedral interstices)

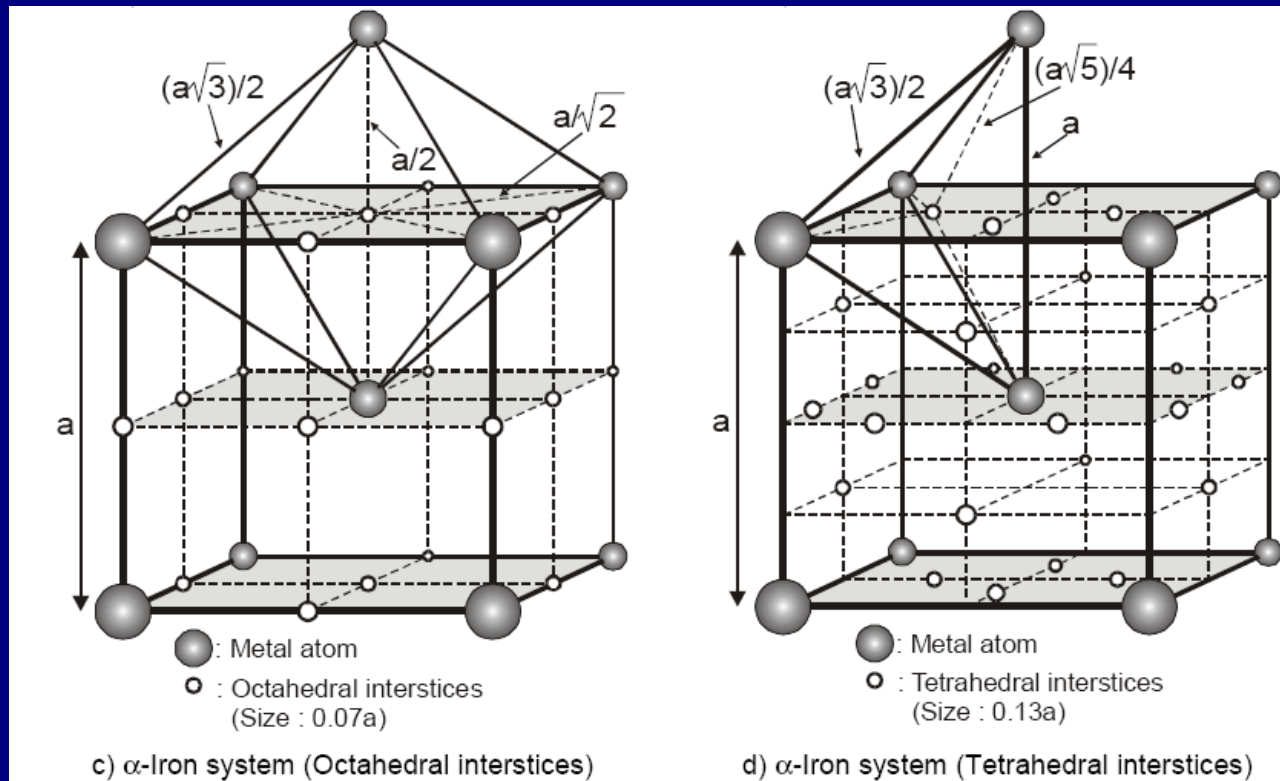


b) γ -Iron system (Tetrahedral interstices)

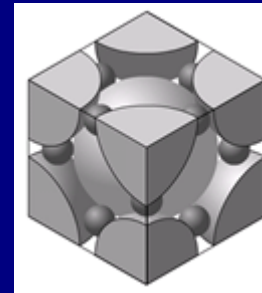
$$\begin{aligned}
 \text{Packing Density} &= \frac{4 \times \left(\frac{4}{3} \pi r^3\right)}{(\sqrt{2} \times r)^3} \\
 &= \frac{\pi}{3\sqrt{2}} \\
 &\approx 74\%
 \end{aligned}$$



Hydrogen interstitial in ferrite



$$\begin{aligned}
 \text{Packing Density} &= \frac{2 \times \left(\frac{4}{3} \pi r^3\right)}{\left(\frac{4}{\sqrt{3}} r\right)^3} \\
 &= \frac{\sqrt{3} \times \pi}{8} \\
 &\approx 68\%
 \end{aligned}$$



Size references



Sizes:

Fe atomic diameter: 2.80 \AA

Mo atomic diameter: 2.90 \AA

Ni atomic diameter: 2.70 \AA

H atomic diameter: 0.50 \AA

BCC unit size: 2.87 \AA

FCC unit size: 11.85 \AA

Ferrite tetrahedral interstice: 0.37 \AA

Austenite octahedral interstice: 1.177 \AA

Austenite spacing in bars microstructure

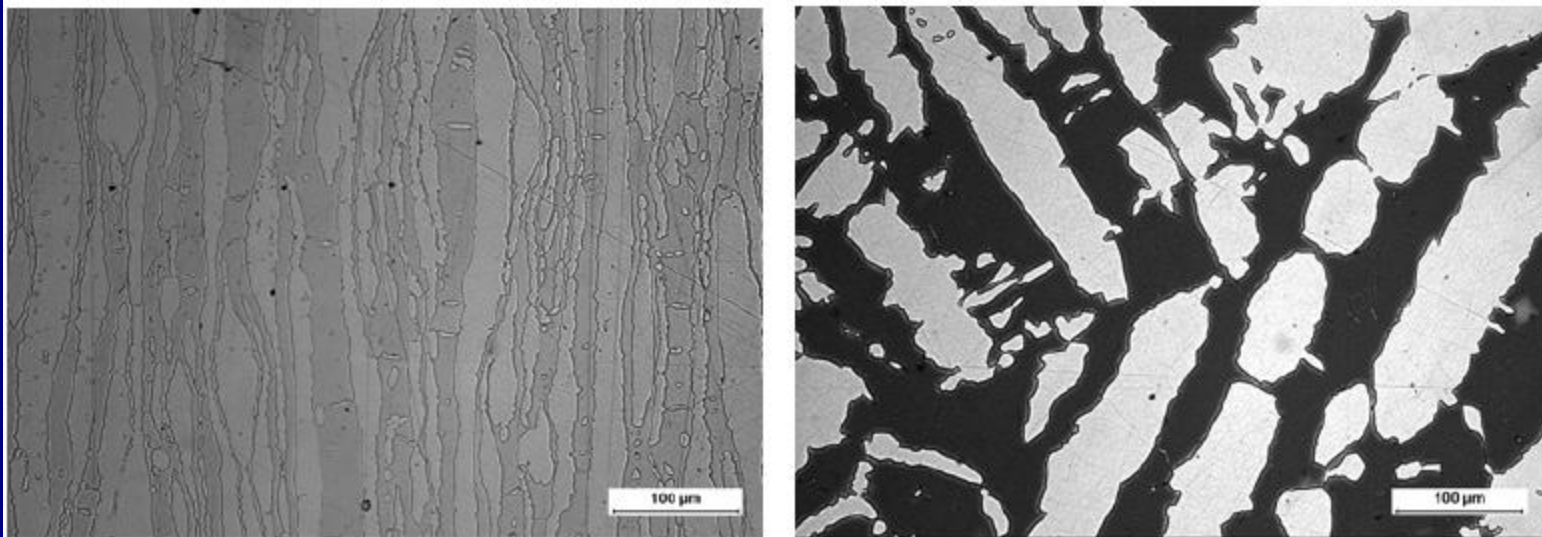


FIGURE 3 – Left, the microstructure of extruded tube (sample 1, 90 % $R_{p0.2\%}$) and right, the microstructure of the large diameter bar (sample 2, 84 % of $R_{p0.2\%}$). The samples are etched in Murakami, which makes the austenite appear white and the ferrite tanned.

Tests at super duplex bars



Test conditions:

- Bar and extruded material of SAF2507 – UNS S32750
- Small test bars and extruded tube with austenite spacing 15–16 μm
- Large test bars with austenite spacing between 32-51 μm
- Pre-charged at 20 mA/cm² in 10% H₂SO₄ with 30 mg/l As₂O₃ as cathodic poison for 24 hours.
- Testing time 500 hours or until failure with constant load relative to yield strength.
- Polarized to -1050 mV SCE at 4°C in artificial sea water 3.5% NaCl
- Testing specimen: standard tensile specimen with total length of 100 mm, gauge length: 25.4 mm diameter 3 mm.
- Constant load testing, exposure time 500 hrs, or until failure

Cracks at super duplex bars

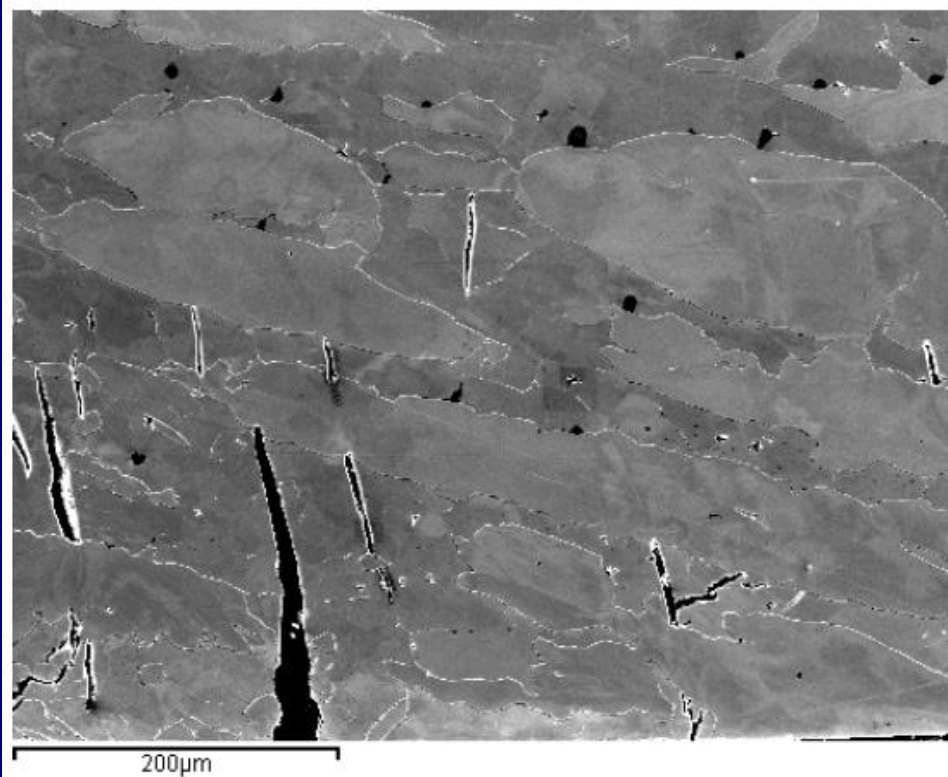


FIGURE 2 – The microstructure of the large dimension bar specimen that failed at 93% of $R_{p0.2\%}(4^{\circ}\text{C})$. The surface of the specimen is at the base of the picture.

Austenite spacing in HIP'ed and forged materials microstructure

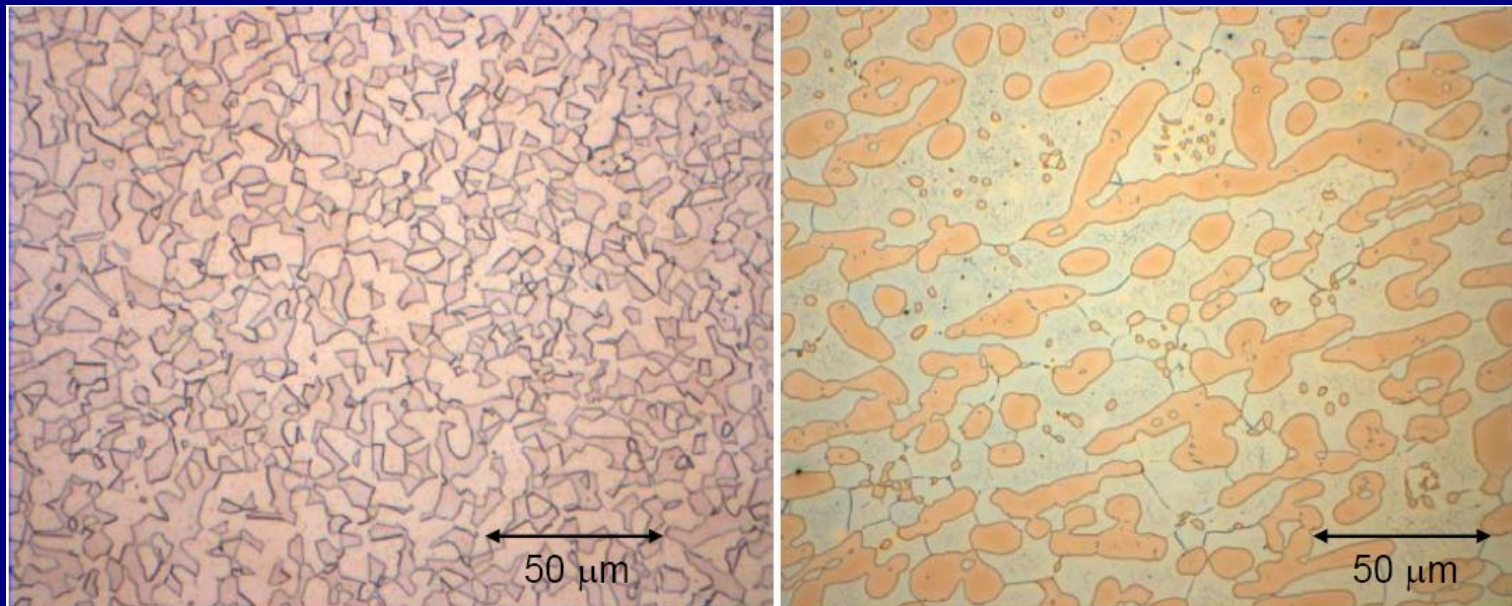


FIGURE 1 - Microstructure of materials: (a) HIP SDSS 32760 and (b) forged SDSS 32760 (M X200).

Grain size: $12.7 \mu\text{m}$

Grain size: $47.9 - 50.9 \mu\text{m}$

Tests at superduplex HIP'ed and forged materials



HIP'ed materials from UNS S31803, UNS S32550 and UNS S32760

Forged materials from UNS S32760

HIP material austenite spacing average 12.7 – 14.6 μm

Forged material austenite spacing average 47.9-50.9 μm

Test temperature: 4°C in artificial sea water 3.5% NaCl

Polarized to -1050 mV SCE

Single edge notch bend testing with crack tip opening displacement, exposure time 30 days.

Crack tip opening displacement (CTOD) test

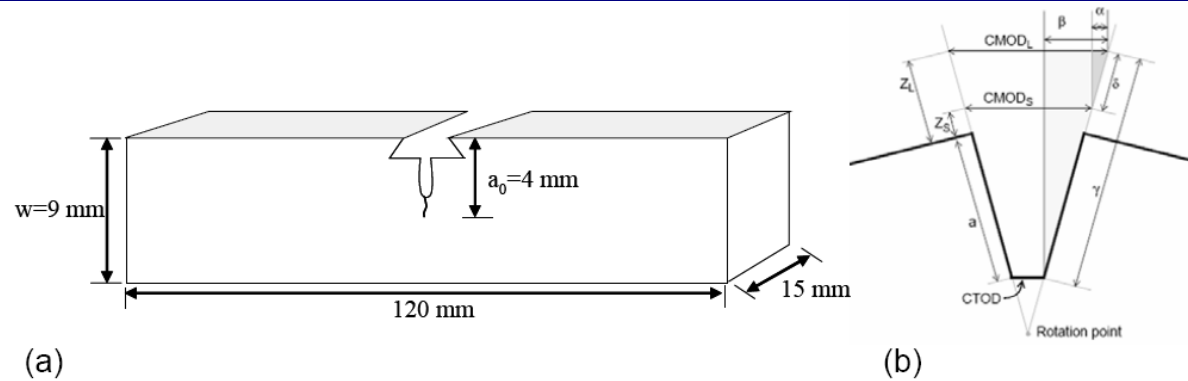


FIGURE 2 - SENB sample dimensions and notch/fatigue pre-crack geometry (a), relationship between measured CMOD and CTOD (b).

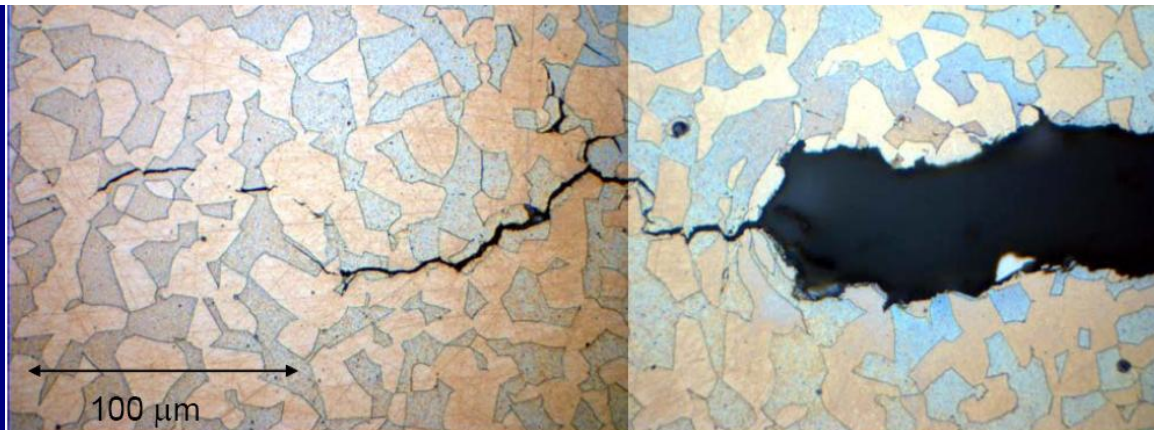


FIGURE 3 - Specimen 32-3(M) SDSS UNS S32760 CTOD=0.08 (Mx500): Crack extends $160\text{ }\mu\text{m}$ from end of fatigue pre-crack, region of cracked ferrite phase extends $280\text{ }\mu\text{m}$ from end of fatigue pre-crack.

Results and conclusion



Results:

1. Large bars with austenite spacing between 38-45 μm failed
2. No failure was detected for materials with austenite spacing 15-16 μm
3. COTD 0.06-0.08 mm for HIP materials and COTD 0.016-0.03 mm for forged materials.

Conclusion

1. The importance of product form and austenite spacing is significant
2. DNV limit of allowable stress max. 80% of yield strength with max. austenite spacing 30 μm is a little bit too conservative that 90% of yield strength is acceptable.
3. For coarser austenite spacing, allowable stress max. 80% of yield strength is reasonable.

Recommended practice: DNV-RP-F112 – October 2008



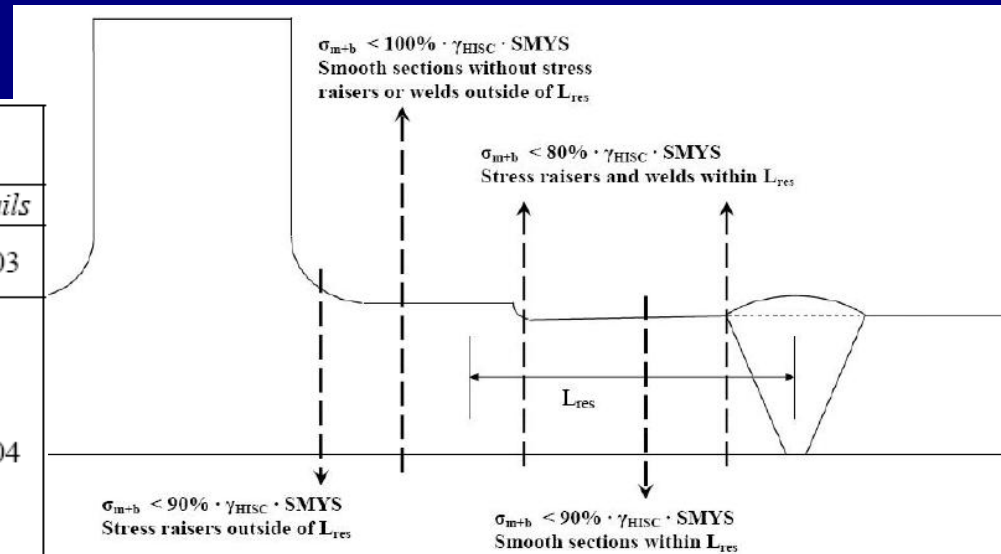
Allowable SMYS factor

Table D1 Allowable SMYS factor for duplex stainless steel
(See also Figure 2)

	α	Area considered	Details
Membrane stress (α_m)	80%	Everywhere	D303
Membrane plus bending stress (α_{m+b})	100%	Smooth sections without stress raiser or welds – outside of L_{res}	D304
	90%	Smooth sections within L_{res}	
	90%	Weld toes attachments (see C403) and stress raisers - outside of L_{res}	
	80%	Weld toes and stress raisers – within L_{res}	

Note:

The value for SMYS at elevated temperatures shall be adjusted for temperature effects, see B100.



$$\sigma_m < \alpha_m * \gamma_{HISC} * SMYS$$

$$\sigma_{m+b} < \alpha_{m+b} * \gamma_{HISC} * SMYS$$

Table D2 HISC material quality factor

(See also C300)

Material	γ_{HISC}	Details
Fine austenite spacing	100%	D305
Coarse austenite spacing	85%	

Recommended practice: DNV-RP-F112 – October 2008



Scope:

This is a recommended practice covers all components made from duplex stainless steel that are installed subsea and are exposed to cathodic protection

Material requirement of duplex/super duplex Stainless steel according to DNV-RP-F112

- Material should be solution annealed and water quenched.
- Material categorized as fine austenite spacing
 - HIP materials
 - Weld metal. HAZ follows the base material.
 - Tube and pipe produce by extrusion, seamless rolling or drawing in all dimensions and wall thickness.
 - Rolled plates with wall thickness < 25 mm
- Materials that do not fall into the category above should be considered have coarse austenite spacing ($> 30 \mu\text{m}$) unless physical measurement of the austenite spacing indicate otherwise.
- Metallographic and material tests:
 - Metallographic characterization of microstructure – ferrite content, inter-metallic phase, austenite spacing.
 - Corrosion test according to ASTM G48
 - Impact test at an appropriate temperature.

References



References:

1. Recommended Practice DNV-RP-F112, “ Design of Duplex stainless steel subsea application equipment exposed to cathodic protection” – October 2008
2. Gro Ostensesn Lauvstad Roy Johnsen ,Bard Nyhus – Sintef , Norway
Martin Bjurstrom and Carl-Gustav Hjorth - Metso Powdermet AB, Sweden. “Improved Resistance towards hydrogen induced stress cracking (HISC) of hot isostatically pressed (HIP) duplex stainless steel under cathodic protection”
3. Sabina Ronneteg, Anna Juhlin and Ulf Kivisakk, AB Sandvik Materials Technology AB - R&D Sandviken, Sweden. “ Hydrogen embrittlement of duplex stainless steels testing of different product forms at low temperature”. Paper 07498 NACE Corrosion, Conference and Expo 2007.
4. Per Olsson, Anna Delblanc Bauer and Hans Eriksson – AB Sandvik Steel R&D center, Sandviken Sweden. “ Hydrogen embrittlement of duplex grades UNS S32750 and UNS S31803 in connection with cathodic protection in chloride solutions”. Presented at Duplex Stainless Steel 97, 5th World conference and Expo, 21-23 October 1997, Maastricht, The Netherland.
5. Ekkarut Viyanit, M.Eng Helmut Schmidt Universitat – Hamburg, Germany, “Numerical Simulation of Hydrogen Assisted Cracking in Supermartensitic Stainless Steel Welds”, Doktor-Ingenieurs dissertation, 2005.